ELECTRICALLY CONDUCTING AND THERMALLY STABLE π -CONJUGATED POLY(ARYLENE)S PREPARED BY ORGANOMETALLIC PROCESSES

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CONTENTS

1.	Introduction	1153
2.	Synthetic procedures	1157
	2.1. Synthesis through Grignard reagent	1157
	2.2. Synthesis using zinc as the reducing reagent for Ni(II) complexes	1164
	2.3. Synthesis by using Ni(0) complexes	1166
	2.4. Synthesis by using Pd-catalyzed coupling reactions between aryl dihalides and	
	diethynylarenes	1174
3.	Structures	1176
	3.1. Structures in the solid and solutions	1176
	3.2. Structures of doped polymers	1183
4.	Properties	1184
	4.1. Thermal stability	1184
	4.2. Chemical properties	1186
	4.3. Electrically conducting properties	1187
	4.3.1. Chemical doping	1187
	4.3.2. Electrochemical doping	1190
	4.4. Optical properties	1193
	4.4.1. UV-visible absorption	1193
	4.4.2. Fluorescence, electroluminescence, and non-linear optical properties	1194
5.	Application	1195
	5.1. Battery	1195
	5.2. Electrochromic display	1198
	5.3. Others	1199
6.	Conclusion and scope	1200
Re	eferences	1201

1. INTRODUCTION

Diorganonickel(II) complexes NiR_2L_n undergo reductive coupling reactions to give R-R, ¹⁻¹³

$$L_n \text{Ni} \stackrel{R}{\swarrow} \rightarrow \text{R-R}$$
 (1)

and this coupling reaction has been utilized to carry out nickel-catalyzed C-C

coupling reactions between Grignard reagent and organic halide (eq. 2), 14-16 and dehalogenation coupling of organic halides with zinc (eq. 3). 17-19

$$RMgX + R'X \xrightarrow{Ni-complex} R-R' + MgX_2$$
 (2)

$$2RX + Zn \xrightarrow{\text{Ni-complex}} R - R + ZnX_2$$
 (3)

In addition to the coupling reactions expressed by eqs 2 and 3, Ullmann type coupling of organic halides using a zero-valent nickel complex itself as a dehalogenation reagent²⁰⁻²⁵

$$2RX + NiL_n \rightarrow R-R + NiX_2 \tag{4}$$

has been developed.

On the other hand, Pd-complexes catalyze the following C-C coupling reactions, which are also considered to proceed through organotransition metal compounds,

$$RC \equiv CH \ + \ R'X \ + \ Et_2NH \xrightarrow{Pd \ complex} RC \equiv CR' \ + \ Et_2NH_2X \quad (5)^{26,27}$$

$$PhC \equiv CMgBr + PhI \xrightarrow{Pd \text{ complex}} PhC \equiv CPh + MgBrI.$$
 (6)²⁸

We have applied the transition metal-promoted C-C coupling found by us to the polycondensation of dihaloaromatic compounds,

Method I

$$nX-Ar-X + nMg \rightarrow [X-Ar-MgX] \xrightarrow{Ni-complex} (Ar)_n$$
 (7)^{29,30}

Method II

$$nX-Ar-X + nZn \xrightarrow{\text{Ni-complex}} (Ar)_n$$
 (8)³¹

Method III

$$nX-Ar-X + nNi(0)L_m \rightarrow (Ar)_n$$
 (9)³²⁻³⁴

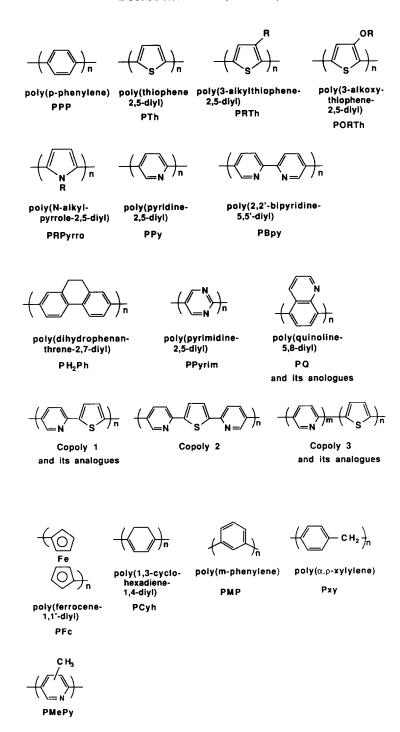
Method IV
$$nHC=C \longrightarrow C=CH + nX-Ar-X$$

$$-(C=C \longrightarrow C=C-Ar)-n + 2n HNEt_3X$$

$$(10)^{35}$$

Method V

to prepare the following electrically conducting and thermally stable π -conjugated aromatic polymers, and related polymers.



1156 T. YAMAMOTO

Among the polymers, PTh, $^{36-41}$ PRTh, $^{39,40,42-44}$ PRPyrro, $^{45-47}$ PPy, $^{32,48-50}$ PBpy, $^{51-53}$ PH₂Phen, 33 PPyrim, 54 PQ⁵⁵ and its analogues, PCyh, 56 Copoly's 1-3, $^{57-59}$ PMePy, 60 and Polyyne $1-6^{35,61}$ were first prepared by the transition metal-promoted polycondensations, Methods I–V, (by us except for PRPyrro). Some π -conjugated poly(arylene)s (e.g. PPP^{29,30} and PFc) with high crystallinity and well-defined bonding between the monomer units were also first prepared by this polycondensation. PTh⁴¹ and PRTh^{43,44} (as materials) as well as the polycondensation Methods I were patented by our institute; PPy³² and PBpy⁵¹ as well as the polycondensation Methods II³¹ and III³⁴ are under patent application.

PTh is the first π -conjugated electrically conducting polymer consisting of 5-membered heteroaromatic rings. It has a π -conjugation system similar to that of poly(acetylene), which has the most simple π -conjugation system and highly electrically conducting properties, but shows only low chemical stability.

Formal substitution of hydrogens of cis-poly(acetylene) affords PTh, which is expected to have high chemical and thermal stability due to the aromaticity of the recurring monomer unit. However, this substitution of H of

poly(acetylene) by S cannot be achieved by usual chemical methods. On the other hand, application of Methods I–III to 2,5-dihalothiophene clearly affords PTh. PTh prepared about 15 years ago^{36,37} still remains chemically unchanged in air at room temperature.

After PTh was reported to have high chemical and thermal stability, 36,37 other various electrically conducting π -conjugated polymers consisting of 5-member heteroaromatic rings (e.g. polypyrrole⁶²) were synthesized.

In this review, we will describe results of the polycondensation, structure of the polymers, and chemical and physical properties of the various π -conjugated polymers shown above.

2. SYNTHETIC PROCEDURES

2.1. Synthesis through Grignard reagent

Various dihalo- as well as polyhaloorganic compounds can be polymerized according to the following reaction (eq. 12).

$$X-R-X + Mg \rightarrow [X-R-MgX] \xrightarrow{\text{transition metal compounds}} (R)_n$$
 (12)

R: arylene or α , ω -alkanediyl. Transition metal: nickel, palladium, iron, chromium, cobalt, copper, etc.

Results of the polymerization of dihaloaromatic compounds are shown in Table 1.^{29,30} As shown in Table 1, nickel compounds serve as the best catalyst among the transition metal complexes tested. When copper complexes are employed, the polymerization expressed by eq. 12 is applicable to the polymerization of α , ω -dihaloalkane (e.g. 1,6-dibromohexane)

$$Br(CH_2)_6Br + Mg \rightarrow [Br(CH_2)_6MgBr] \xrightarrow{Cu \text{ complex}} (CH_2)_{6n}$$
 (13)

to afford poly(methylene) or poly(ethylene) (Mw = ca. 5000), ⁶³ which shows very sharp v(C-H) absorption bands comparable to those of Ziegler-Natta poly(ethylene).

The polymerization proceeds very smoothly at room temperature and at warm (e.g. reflux temperature of THF) conditions. Analysis of the products of the reaction of X–R–X and Mg revealed that the reaction of Br– C_6H_4 –Br mainly affords mono-Grignard reagent Br– C_6H_4 –MgBr, ³⁰ whereas those of Br(CH₂)₆Br and 2,5-dibromothiophene yield a 1:1 mixture of Br(CH₂)₆Br and BrMg(CH₂)₆MgBr⁶³ and a 1:2:1 mixture of $C_4H_2SBr_2$, C_4H_2S (MgBr)Br, and C_4H_2S (MgBr)₂, ³⁹ respectively.

$$Br-C_6H_4-Br + Mg \longrightarrow Br-C_6H_4-MgBr$$
 (14)

$$Br(CH_2)_6Br + Mg \longrightarrow 1:1 \text{ mixture of } Br(CH_2)_6Br$$

and $BrMg(CH_2)_6MgBr$ (15)

Br-
$$\sqrt{s}$$
-Br + Mg \longrightarrow 1:2:1 mixture of C₄H₂SBr₂, C₄H₂S(MgBr)Br, and C₄H₂S(MgBr)₂ (16)

The polymerization can also be carried out by using pseudo halides such as tosylates, 63

$$Y-R-Y \xrightarrow{Organometallic process} polymer (Y = halogen, OTs, etc.).$$

TABLE 1. Dehalogenation polymerization of haloaromatic compounds with magnesium in the presence of transition metal catalysts*

No.	Monomer	Catalyst†	Solvent‡	Time (h)	Yield§ (%)	Remarks
1	p-Dibromobenzene	none	THF	4.0	0	
2	<i>p</i> -Dibromobenzene	NiCl ₂ (bpy)	THF	4.0	95	
3	p-Dibromobenzene	NiCl ₂ (bpy)	THF	6.0	94	
4	p-Dibromobenzene	NiCl ₂ (bpy)	Bu ₂ O	4.5	68	
5	<i>p</i> -Dibromobenzene	$NiBr_2(PPh_3)_2$	THF	4.0	64	
6	p-Dibromobenzene	PdCl ₂ (bpy)	THF	4.0	95	
7	p-Dibromobenzene	FeEt ₂ (bpy) ₂	THF	6.0	18	
8	p-Dibromobenzene	CrMrCl ₂ (THF) ₃	THF	7.0	11	
9	p-Dibromobenzene	Ni(cod)(PPh ₃) ₂	THF	4.0	86	
10	p-Dibromobenzene	NiCl ₂	THF	4.0	83	
11	p-Dibromobenzene	FeCl ₂	THF	4.0	76	
12	p-Dibromobenzene	FeCl ₃	THF	4.0	68	
13	p-Dibromobenzene	CoCl ₂	THF	4.0	77	
14	p-Dibromobenzene	NiCl ₂ (bpy)	THF	6.0	60	
15	p-Diiodobenzene	NiCl ₂ (bpy)	THF	4.0	50	
16	m-Dichlorobenzene	NiCl ₂ (bpy)	THF	6.0	92	
17	Bis(p-bromophenyl)ether	NiCl ₂ (bpy)	THF	5.0	60	
18	Bis(p-bromophenyl)ether	$NiBr_2(PPh_3)_2$	THF	9.0	39	
19	α,ρ-Dichlorotoluene	NiCl ₂ (bpy)	THF	7.0	69	
20	2,5-Dibromopyridine	NiCl ₂ (bpy)	THF	8.0	62	
21	2,5-Dibromopyridine	NiBr ₂ (PPh ₃) ₂	THF	8.0	64	
22	2,5-Dibromopyridine	NiCl ₂	THF	8.0	23	
23	9,10-Dibromoanthracene	NiCl ₂ (bpy)	THF	6.0	22	
24	1,3,5-Trichlorobenzene	NiCl ₂ (bpy)	THF	15		
25	1,3,5-Trichlorobenzene	NiCl ₂ (bpy)	Diglyme	10.5	72	
26	Hexachlorobiphenyl	NiCl ₂ (bpy)	THE	15.0		
27	2,5-Dibromothiophene	NiCl ₂ (bpy)	THF	5.0	78	
28	2,5-Dibromothiophene	NiCl ₂ (bpy)	$THF + n-Bu_2O$	5.0	48	
	_		(1:4)			
29	2,5-Dichlorothiophene	NiCl ₂ (bpy)	THF	4.0	40	
30	3-Methyl-2,5-dibromothiophene	NiCl ₂ (bpy)	THF	8.0	86	
31	3-Methyl-2,5-dibromothiophene	NiCl ₂ (dpp)	THF	10.0	79	
32	2,4-Dibromothiophene	NiCl ₂ (bpy)	THF + n -Bu ₂ O	5.0	58	
			(1:4)			

^{*} Under reflux except for No. 3.

[†] bpy = 2.2'-bipyridine; PPh₃ = triphenylphospine; cod = 1.5-cyclooctadiene; dpp = 1.3-bis-diphenylphosphino)propane; Me = methyl; Et = ethyl.

[‡] THF = tetrahydrofuran, Bu₂O = dibutyl ether under reflux. § Based on the amount of carbon taken into the polymer.

At 0°C.

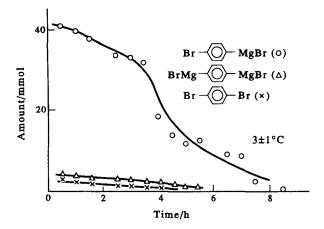


FIG. 1. Time course of the polymerization of p-C₆H₄Br₂ (50 mmol) monitored by consumption of monomers. On THF (50 ml) at 3 ± 1 °C. O: p-BrC₆H₄MgBr; \triangle : p-C₆H₄(MgBr)₂; \times : p-C₆H₄Br₂. Catalyst = NiCl₂ (bpy) 50 mg).

The time course of the polymerization of $Br-C_6H_4-Br$, as monitored by the consumption of the chemical species at low temperature (3°C), is shown in Fig. 1.

Color and thermal stability of poly(arylene)s prepared by this method are shown in Table 2. As shown in Table 2, π -conjugated PPP, PTh and PRTh show absorption in the visible region due to the π - π * transition, whereas non-conjugated PMP, $(-C_6H_4OC_6H_4)_n$, $(-C_6H_4CH_2)_n$, and poly(thiophene-2,4-diyl)

$$\left(\left\langle \right\rangle \right)_{n}$$

are white or colorless.

In the case of poly(acetylene), introduction of a substituent usually affords white or colorless $\{CH=CR\}_n$ in which the π -conjugation is essentially destroyed due to the steric effect of the R group. On the other hand, introduction of a substituent in the five-membered-ring π -conjugated polymers does not cause the destruction of the π -conjugation, due to the presence of less steric repulsion between the R group and the five-membered ring.⁴²

1160

No.*	Polymer	Color	Mp (°C)
2	$+p$ -C ₆ H ₄ \rightarrow_{p} (PPP)	light yellow	550 dec
6	$+p-C_6H_4$	light yellow	550 dec
14	$+p-C_6H_4$	light yellow	550 dec
17	$+m$ - C_6H_4 $\frac{n}{n}$ (PMP)	white	190-200† 280-295‡
18	$+C_6H_4CH_2$	white	230-235† > 400‡
26	+ C ₆ H ₃ +	brown	290 dec
28	poly(thiophene-2,5-diyl) (PTh)	reddish brown	116§ ca. 320∥
31 33	poly(3-methylthiophene-2,5-diyl) (PRTh) poly(thiophene-2,4-diyl)	reddish brown white	ca. 300 ca. 320

TABLE 2. Properties of poly(arylene)s

- * Numbers correspond to those in Table 1.
- † For fraction soluble in hot toluene.
- [‡] For fraction insoluble in hot toluene.
- § For fraction soluble in hot CHCl₃ softening point.
- || For fraction insoluble in hot CHCl₃.

Based on this finding, a number of 3-substituted poly(thiophene-2,5-diyl)s^{39,40,42-44,64-66} and poly(pyrrole-2,5-diyl)s⁶⁷⁻⁷⁰ have been synthesized, and their electrical, chemical, and physical properties have been investigated.

The basic condensation reaction (eq. 2) for the polycondensation (eq. 12) is considered to proceed through the following organonickel-mediated catalytic cycle involving: (a) formation of NiR_2L_2 by the reaction of NiX_2L_2 with RMgX; (b) reaction of NiR_2L_2 with R'X to yield R-R and $NiR'(X)L_2$; (c) alkylation of $NiR'(X)L_2$ with RMgX to yield $NiR(R')L_2$; and (d) elimination of R-R' with formation of $NiR'(X)L_2$ in the reaction of $NiR(R')L_2$ with R'X. 14,30

In the mechanism the starting complex NiX_2L_2 is converted into the active species $NiR'(X)L_2$ by reactions (a) and (b). The catalytic formation of R-R' then proceeds through the key intermediates $NiR'(X)L_2$.

As for the coupling of R and R' groups on Complex 2 (reaction d), coordination of the aromatic halides R'X to Ni

may take place to facilitate the coupling reaction, since coordination of a π -accepting ligand on diorganonickel(II) complexes strongly enhances the reductive coupling of the organic groups on Ni.¹

If the same mechanism shown above can be applied to the present polymerization of p-C₆H₄Br₂, it seems to proceed as follows: where P(n) stands for the polymer having n phenylene units and P(1) represents the monomer p-BrC₆H₄MgBr which is the main product of the reaction between p-C₆H₄Br₂ and Mg. 1(n_1), 2(n_1 , n_2).... represent intermediate nickel complexes 1 and 2 having phenylene groups bonded to nickel.

Figures 2 and 3 show IR spectra of some of the prepared poly(arylene)s, and Fig. 4 shows X-ray diffraction patterns of some of the poly(arylene)s. The IR spectrum of poly(p-phenylene) prepared by the present method is essentially the same as those of poly(p-phenylene)s prepared by other methods such as oxidative cationic polymerization of benzene (Kovacic's method^{71,72}). The out-of-plane vibration of p-phenylene units, $\delta(para)$, in poly(-phenylene)s prepared

SCHEME 1.

1162 T. YAMAMOTO

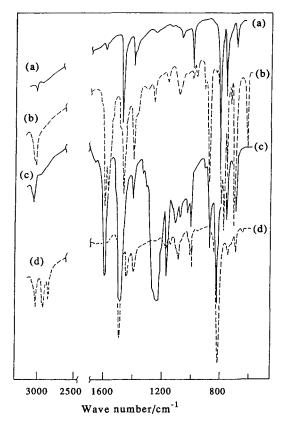


Fig. 2. IR spectra of (a) poly(p-phenylene) (PPP) prepared from p-C₆H₄Br₂; (b) poly(m-phenylene) (PMP); (c) poly(oxybiphenylene); (d) poly(phenylenemethylene).³⁰

under various conditions appears in the region $805 \pm 5 \,\mathrm{cm}^{-1}$. Weak v(C-Cl), v(C-Br), and v(C-I) bands of terminal p-halophenyl groups are observed at 1085, 1065, and $1055 \,\mathrm{cm}^{-1}$, respectively, in the IR spectra of poly(p-phenylene)s prepared from the corresponding p-dihalobenzenes. In addition to the $\delta(para)$ band at $805 \pm 5 \,\mathrm{cm}^{-1}$ assignable to the p-phenylene units, the IR spectrum of poly(p-phenylene) shows two out-of-plane vibration bands due to terminal phenyl groups at $760 \,\mathrm{cm}^{-1}$ ($\delta(\text{mono}_1)$) and $690 \,\mathrm{cm}^{-1}$ ($\delta(\text{mono}_2)$). These IR spectroscopic data suggest that the poly(p-phenylene) has both p-halophenyl and phenyl groups as the terminal groups:

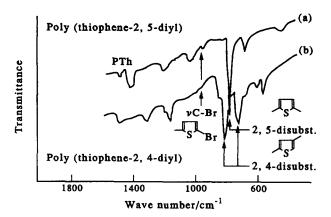


Fig. 3. IR spectra of (a) poly(thiophene-2,5-diyl) PTh; and (b) poly(thiophene-2,4-diyl) prepared from 2,5-dibromothiophene and 2,4-dibromothiophene, respectively.³⁹

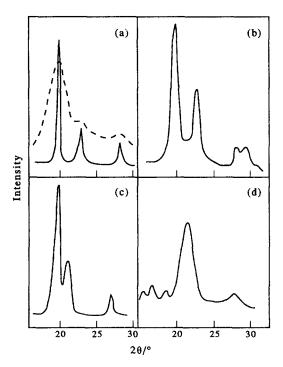


Fig. 4. Powder X-ray diffraction patterns (CuK₂) of (a) poly(*p*-phenylene) (PPP); (b) poly(oxybiphenylene); (c) poly(phenylene-methylene); and (d) poly(*m*-phenylene) PMP.³⁰ Data of PPP prepared by oxidation of benzene by CuCl₂ in the presence of AlCl₃^{71,72} are also shown by a broken line in (a).

The fact that the intensities of the bands due to the terminal phenyl group $(\delta(\text{mono}_1) \text{ and } \delta(\text{mono}_2))$ are much greater than that of v(C-X) band suggests that the proportion of the terminal group bearing the halogen is not high.

Kovacic and Oziomek took the relative intensity of $\delta(para)$ to that of $\delta(\text{mono}_1)$, and $\delta(\text{mono}_2)$, $A(\delta(para))/\{A(\delta(\text{mono}_1)) + A(\delta(\text{mono}_2))\}$ (A = absorbance of the peak), as a measure for the degree of polymerization. The value $A(\delta(para))/\{A(\delta(\text{mono}_1)) + A(\delta(\text{mono}_2))\}$ for poly(p-phenylene) prepared from dibromobenzene with various catalysts lies between 1.92 (No. 2, Table 1) and 3.85 (No. 7, Table 1). This indicates that poly(p-phenylene)s obtained by this method have relatively high degrees of polymerization, comparable to those of polymers prepared by Kovacic and Oziomek through oxidative cationic polymerization of benzene.

The IR spectrum of PTh gives rise to a different absorption pattern from that of poly(thiophene-2,4-diyl) in the out-of-plane $\delta(CH)$ vibration region (Fig. 3). PTh can be also prepared through electrochemical^{73,74} and chemical^{74,75} oxidation of thiophene, and PTh's thus prepared may contain some of the thiophene-2,4-diyl unit, as shown by comparison of their IR spectra with those shown in Fig. 3.

PPP prepared by dehalogenation condensation gives rise to three main X-ray diffraction bands with good sharpness between $2\theta=18^{\circ}$ and 30° (CuK_a). The three bands are regarded as associated with the diffractions from the *ab* planes (*c* axis is the direction of the polymer chain). PTh affords three main X-ray diffraction bands at almost the same 2θ values as those of PPP, and two-dimensional structures with the same *ab* parameters. Honorealed PPP prepared by oxidation of benzene with CuCl₂-AlCl₃ (Kovacic's method) gives rise to somewhat broad X-ray diffraction bands (cf. broken line in Fig. 4a). Preparation of PPP in the presence of carbon monoxide gives PPP with some ketone groups in the chain.

2.2. Synthesis using zinc as the reducing reagent for Ni(II) complexes

Table 3 summarizes results of Ni-catalyzed dehalogenation polycondensation of 2,5-dibromothiophene, 3-methyl-2,5-dibromothiophene, and *p*-diiodobenzene with zinc.³¹ Palladium compounds also show

$$nX-Ar-X + nZn \xrightarrow{Ni- \text{ or Pd-catalyst}} (Ar)_n$$
 (17)

similar catalytic activity (run 3). PTh, PRTh ($R = CH_3$) and PPP thus prepared have essentially the same structures as those of the polymers prepared by other

TABLE 3. Ni- or Pd-catalyzed dehalogenation polycondensation of dihaloaromatic compounds with zinc31

	mount in mmol)	Catalyst (amount in mmol)	Amount of Zn in mmol	Additive (amount in mmol)	Solvent†	Temp.	Time (h)	Yield‡
_	C ₄ H ₂ SBr ₂ (18)	NiCl ₂ (dpe) (0,2)	54	None	HMPA	150	2	78
2	_	NiBr, (PPh,), (1,8)	54	NEt ₄ 1 (36)	DMF	100	∞	30
3	$\overline{}$	PdCl, (0,7)	65	$NEt_4 I (44) + PPh_3 (3)$	Benzene	80	70	09
4	C, H, SBr, (42)	$NiBr_2(PPh_1)$, (1,3)	65	$NEt_4 I (87)$	Benzene	80	5	62
2	C ₆ H ₄ I ₂ (11)	$NiBr_2(PPh_3)_2(0,5)$	30	$NEt_4 I (20) + PPh_3 (2)$	DMF	150	18	99

* C₄H₂SBr₂, 2,5-dibromothiophene; C₅H₄SBr₂, 3-methyl-2,5-dibromothiophene; C₆H₄l₂, 1,4-diiodobenzene. † HMPA, hexamethylphosphoric triamide; DMF, N.N-dimethylformamide. ‡ Based on carbon recovered.

methods, as revealed by IR and X-ray diffractometry. Since the basic C-C coupling reaction (eq. 3) is applicable to organic halides with functional groups such as -COR and -CN and to reaction systems using polar solvents such as HMPA and DMF, the polycondensation can also be carried out by using 2,5-dibromothiophene with -CHO, -CH=NOH, or -CN substituents at the 3-position, which are sensitive to the Grignard reagent and cannot be polymerized by the X-R-MgX mediated method (eq. 12). However the Mn values of the products obtained using these unique monomers, so far, have not been high (Mn = 520-630). On the other hand, Ueda and his coworkers applied the Ni-catalyzed polycondensation using Zn, to the preparation of aromatic polyether-polyketones and poly(3-phenyl-2,5-thienylene) PRTh ($R = C_6H_5$) from the corresponding organic dihalides. Their polymers had $[\eta]$ values of 0.50-0.87 dl g^{-1} .

2.3. Synthesis by using Ni(0) complexes

Since coupling reaction 4 proceeds under mild conditions, it can be applied to a wide range of aromatic compounds (e.g. those with carbonyl and cyano group) under various reaction conditions (e.g. in various solvents), and is the most direct and simple reaction among the Ni-based coupling reactions. Method III, expressed in reaction 9, is expected to provide a versatile means for molecular design and synthesis of electrically conducting π -conjugated polymers.

Polycondensation of the following dihaloorganic compounds affords π -conjugated polymers with well-defined linkages between the monomer units, to be suited to the investigation of crystal structure and doping processes of the polymers.³³

Table 4 summarizes results of polymerization of 2,5-dihalothiophene, 1,4-dibromobenzene and related monomers. As the dehalogenating Ni(0)-complex, mixtures of bis(1,5-cyclooctadiene)nickel, Ni(cod)₂, and neutral ligand L were primarily employed.³³

As shown in runs 1-6 in Table 4, the dehalogenation polycondensation of $2,5\text{-Br}_2\text{-Th}$ with a mixture of $\text{Ni}(\text{cod})_2$ and PPh_3 proceeds smoothly in the temperature range 25 through 100°C . On mixing $2,5\text{-Br}_2\text{-Th}$ with the Ni(0)-complex, precipitation of reddish brown polymer starts immediately, and the best yield is attained at $60\text{--}80^{\circ}\text{C}$. The polymer obtained has the following π -conjugated PTh structure as revealed by its analytical data and comparison of its IR spectrum with that of the above described PTh. Most of the obtained

$$nx - x + nNiL_m$$

$$2.5-X_2-Th$$

$$s$$

$$pTh$$

$$(18)$$

PTh was unextractable by hot CHCl₃, indicating that most of the PTh had a molecular weight higher than 3000.³⁹ Even at short polymerization time (2 h, run 6), a quantitative yield is almost attained. However, PTh obtained after short polymerization time had a relatively high content of Br (3.9%), and the content of Br decreased to 1.8% (at 8 h at 60°C) and 0.7% (at 16 h at 60°C, run 3) with increasing polymerization time, revealing that the chain growth by coupling of the seemingly insoluble precipitate of PTh having a terminal C-Br bond, proceeded even after the formation of the precipitate. The CHCl₃-extractable fraction of PTh decreased from 3.2 to 1.2% with increase in the polymerization time from 2 to 16 h at 60°C.

Ni(PPh₃)₄ also affords PTh (run 7), however the yield is lower, indicating that the coordinatively saturated Ni(PPh₃)₄ has lower reactivity compared with the mixture of Ni(cod)₂ and PPh₃. Use of bpy, instead of PPh₃, affords similar results (runs 8 and 9). 2,5-Cl₂-Th shows lower reactivity than 2,5-Br₂-Th to afford lower yield (run 10). The CP-MAS ¹³C-NMR spectrum of the present PTh (solid) shows two peaks at 136 and 125 ppm assigned to 2,5-carbon and 3,4-carbon atoms, respectively, and no other peak was observed.

Polymerization of 2,5-dihalo-3-alkylthiophenes, 2,5- X_2 -RTh (R = CH₃, n-C₆H₁₃, n-C₈H₁₇, n-C₁₂H₂₅), gives the corresponding soluble poly(3-alkylthiophene-2,5-diyl)s PRTh's, ^{39,40,42-44, 64-66} which, in spite of the sterically large R group, are considered to have π -conjugation systems similar to that of PTh as judged from their reddish brown color and electrically conducting property. ⁴² On the other hand, in the case of substituted poly(acetylene)s +CH=CR + π , the π -conjugation of poly(acetylene) is lost by substitution with R, and the polymer is generally white. CPK molecular models explain the difference in the results of

1168

Table 4. Preparation of poly(thiophene-2,5-diyl), poly(p-phenylene) and related polymers by dehalogenation polycondensation of X-Ar-X with mixture of Ni(cod), and neutral ligand L³³

Run	Monomer (X-Ar-X)*	Ligand†	React	Reaction conditions		Yield‡	Molecular weight§	weight§
			Solvent	Temp. (°C)	Time		M_n	Mw
_	2,5-Br,-Th (0.25)	PPh,	DMF	100	16	92		
7	2,5-Br ₂ -Th (0.25)	PPh	DMF	80	16	100		
33	2,5-Br ₂ -Th (0.25)	PPh	DMF	9	16	100		
4		PPh	DMF	40	16	80		
5		PPh	DMF	25	24	74		
9	2,5-Br ₂ -Th (0.25)	PPh	DMF	99	2	100		
7	$2,5-Br_2-Th (0.25)$	Ni(PPh,),	DMF	09	16	63		
∞	2,5-Br ₂ -Th (0.25)	bpy	DMF	09	16	91		
6	2,5-Br ₂ -Th (0.25)	bpy	DMF	25	24	93		
10		PPh	DMF	9	16	55		
11	2,5-Br ₂ -3MeTh (0.25)	PPh_{j}	DMF	09	16	92		
12	2,5-1 ₂ -3HexTh (0.13)	PPh	DMF	09	48	99		
13	$2,5-I_2-3$ HexTh (0.25)	pby	DMF	9	16	70		52,000
							(74,000)	(30,300)
14	$2,5-I_2-3$ HexTh (0.13)	bpy	DMF	9	84	46		190,000
15	$2,5-I_2-3$ HexTh (0.18)	bpy	toluene	9	48	09		
16	$2,5-I_2-3$ OctTh (0.13)	bpy	DMF	09	48	80		
				10				

(000, 00)											
08	87	91	09	26	66	95	91	92	95	0 (complex)	06
								16			
30	09	09	25	09	09	30	09	09	09	09	99
DMF	DMF	DMF	DMF	DMF	DMF	DMF	DMF	DMF	DMF	THF	DMF
bpy	hpy	PPh,	ppy	PPh	pby	bpy	vad	bpy	bpy	PPh	bpy
2.5-L,-3OctTh (0.13)	2.5-1,-3DodTh (0.13)	$2.5-Br_3-3CNTh (0.25)$	2.5-Br,-3CNTh (0.25)	1.4-BrBen (0.20)	1.4-BrBen (0.20)	1.4-BrBen (0.20)	1.4-L-Ben (0.20)	1,4-Cl,-Ben (0.20)	4.4'-Br ₂ -diBen (0.20)	9.10-X,-Anth (0.10)	$2,7-Br_2-9,10-H_2$ Phen (0.13)
17	<u>×</u>	6 6	30	2 5	: 2	23	2 4	. 52	36	27	28

* 2,5-X₂-Th = 2,5-dihalothiophene; 2,5-I₂-3RTh = 2,5-diiodo-3-alkylthiophene (R: Hex = hexyl; Oct = octyl; Dod = dodecyl); 2,5-Br₂-CNTh = 2,5-dibromo-3-eyanothiophene; 1,4-X₂-Bz = 1,4-dihalobenzene; 2,7-Br₂-9,10-H. Phen = 2,7-dibromo-9,10-dihydrophenanthrene

† PPh₃ = triphenylphosphine; bpy = 2.2-bipyridyl.

‡ Yield was calculated based on carbon recovered: (% carbon in polymer) × (weight of polymer)/(weight of carbon in monomer).

§ Mn = number average molecular weight; Mw = weight average molecular weight. The values without parentheses determined by a light scattering method, whereas the values in parentheses were determined by gel permeation chromatography (GPC).

|| Ni(PPh₃)₄ was used instead of the mixture of Ni(cod)₂ and L.

substitution on poly(acetylene) and on PTh, containing a 5-membered ring.⁴²

$$n_{X} = \prod_{s=1}^{R} x + n_{NiL_{m}}$$
PRTh

PRTh has been prepared by polycondensation *via* a Grignard intermediate, ^{42-45,64} oxidation of 3-alkylthiophene, ⁸³⁻⁸⁵ and electrochemical polymerization. ⁸⁶⁻⁹⁰ Comparison of ¹H-NMR spectra ⁹⁰ of the variously prepared PRTh's reveals that the content of head-to-head junctions in PRTh prepared by the organometallic technique (methods using Ni(0) complex or the Grignard intermediate)

is considerably higher (ca. 60–70%) than that (ca. 10%) in PRTh's prepared by oxidation and electrochemical polymerization. In the ¹H-NMR spectrum of the present PRTh, the peak of the thiophene ring-attached-CH₂ protons $\delta 2.55 \text{ ppm}$)⁹⁰ of the head-to-head unit is stronger than that ($\delta 2.8 \text{ ppm}$) of the head-to-tail unit.

The high content of the head-to-head unit in PRTh prepared by the organometallic techniques suggests that the organometallic coupling reactions (eqs 2-4) proceed with some regio-selectivity; presumably the following 4,4'-dialkyl-5,5'-dihalo-2,2'-bithiophene is the major species formed in the initial step of the polycondensation.

$$2 \times \frac{1}{5} \times$$

Easier oxidative addition of the 5-C-X bond to Ni(0)-complex than that of 2-C-X occurs because of the steric effect of R which explains the major formation of the symmetrical bithiophene derivative shown above, and thus explains the high content of the head-to-head junction in PRTh. Increase in polymerization time (cf. runs 13 and 14 in Table 4) leads to increase of molecular weight of poly(3-hexylthiophene-2,5-diyl). The weight-averaged molecular weight Mw of PRTh determined by the light scattering method roughly agrees with that determined by GPC method (polystyrene standard, cf. run 13). PRTh shows a π - π * absorption band at 380-400 nm in CHCl₃. Both the present PTh and PRTh have good thermal stability; the TGA analysis under N₂ shows residual weight of 80% at 600°C and 81% at 400°C for PTh and PRTh (R = n-C₁₂H₂₅), respectively. Polymerization of 2,5-Br₂-CNTh (runs 19 and 20) gives the corresponding poly(3-cyanothiophene-2,5-diyl) whose IR spectrum shows v(C \equiv N) vibration at 2200 cm⁻¹. This polymer is sparingly soluble in organic solvents.

Polymerization of 1,4- X_2 -Ben and 4,4'-Br₂-diBen affords yellow and fluorescent PPP in high yield when it is carried out in the presence of bpy (runs 22–26). However, carrying out the polymerization in the presence of PPh₃ (run 21) affords PPP in only low yield. The difference between the two polymerizations may be attributed to the difference in the configuration of the intermediate nickel(II) complexes formed by using bpy and PPh₃. The polymerization is now assumed to proceed through oxidative addition of 1,4- X_2 -Ben to a Ni(0) complex to form a (4-halophenyl)halonickel(II) complex L_m NiX(C_6 H₄X), disproportionation of L_m NiX(C_6 H₄X) to afford a di(4-halphenyl)nickel(II) complex L_m Ni(C_6 H₄X)₂, and reductive elimination of X_6 H₄C₆H₄X from L_m Ni(C_6 H₄X)₂. A propagation process involving coupling between X(C_6 H₄)_XX and X(C_6 H₄)_YX is also accounted for by similar basic reactions.

1,4-
$$X_2$$
-Ben + NiL_m $\xrightarrow{\text{oxidative}}$ L_m NiX(C₆H₄X) (21)

$$\xrightarrow{\text{disproportionation}} \text{Ni}(C_6 H_4 X)_2 L_m + \text{Ni}X_2 L_m$$
 (22)

reductive (23)
$$\begin{cases}
\text{reductive} \\
\text{elimination}
\end{cases}$$

$$X-C_6H_4-C_6H_4-X$$

Such basic reactions shown above have been well established in organometallic chemistry, $^{1-13,25}$ and it is known that tertiary phosphine ligands like PPh₃ usually afford trans-Ni(II) complex 91,92 whereas the bidentate bpy ligand definitely affords cis-Ni(II) complex. The reductive elimination usually proceeds more easily in the cis-type complex than the trans-type complex, $^{1-13,93-96}$ and the disproportionation reaction involving exchange of the anionic ligands X and C_6H_4X (eq. 22) also may proceed more easily in the cis-type complex.

In the case of $9,10-X_2$ -Anth (run 27), its reaction with a mixture of Ni(cod)₂ and PPh₃ affords a complex corresponding to L_m NiX(C_6H_4X), Complex 1, in high yield, and it has been revealed

$$X - X + Ni(cod)_2 + PPh_3$$

$$X - Ni - X$$

$$PPh_3$$

$$PPh_3$$

$$PPh_3$$

$$Complex 1a (X = Br)$$

$$Complex 1b (X = Cl)$$

that the 10-C-X bond in Complex 1 is inert toward excess Ni(0) complex, presumably due to the steric hindrance of the bulky anthryl group and the decrease in electrophilicity of the 10-carbon in Complex 1, caused by back donation from nickel. Complex 1 has high thermal and chemical stability; it is stable not only in boiling toluene under air for several hours but also in hydrochloric acid (2 m) in a dispersion system. In the case of Complex 1, the steric effect⁹¹ of the bulky anthryl group as well as partial formation of double bonds between nickel and the aromatic group⁹⁷ accounts for the high stability of the Ni-carbon bond. Due to the extremely high stability of the Ni-carbon bond, 9,10-X₂-Anth does not afford the polymer (run 27). Addition of bpy, instead of PPh₃, as the neutral ligand does not afford the polymer, either. Use of 2,7-Br₂-9,10-H₂Phen as the monomer (run 28) yields a yellow fluorescent polymer, PH₂Phen, which is partly soluble in CHCl₃ and N-methylpyrrolidone (NMP).

The yellow color and the fluorescent property support that the obtained

polymer has a π -conjugation system similar to that of PPP. The IR spectrum of the polymer shows peaks of two categories, one originating from the PPP-like structure (e.g. $\delta(CH)$ at $814\,\mathrm{cm}^{-1}$) and another from the ethylene unit (e.g. v(CH) at 2930, 2890, and $2830\,\mathrm{cm}^{-1}$). Use of BrCH=CHBr in combination with the mixture of Ni(cod)₂ and neutral ligand (PPh₃ or bpy) gives a black solid, whose IR spectrum is similar to that of *trans*-poly(acetylene), showing a characteristic sharp absorption peak at $1010\,\mathrm{cm}^{-1}$, whereas an attempt to polymerize BrCH=CHBr by nickel-catalyzed polycondensation with Mg (eq. 7) leads only to evolution of gas (presumable acetylene). Since the Ni(0) complex can be generated by electrochemical reduction of the Ni(II) complex, and the Ullmann type C-C coupling (eq. 4) can be achieved using the electrochemically generated Ni(0)-complex, the present polymerization can also be carried out electrochemically starting from the dihaloaromatic compounds as the monomer and Ni(II) complexes (e.g. [Ni(bpy)₃]Br₂) as the catalyst; details of the electrochemical polymerization will be reported elsewhere.

The PPP prepared by using the Ni(0) complex seems to have higher molecular weight than the PPP's prepared by the Ni-catalyzed Grignard-reagent mediated dehalogenation polycondensation of 1,4- X_2 -Ben (eq. 7) and by the Kovacic method, 71,72 as judged from its IR spectrum and relatively low bromine content. The IR spectrum of the present PPP shows out-of-plane vibration at $804 \pm 2 \, \mathrm{cm}^{-1} \, (\delta(para))$ and those of the terminal phenyl unit at $760 \, (\delta(\mathrm{mono_1}))$ and $690 \, \mathrm{cm}^{-1} \, (\delta(\mathrm{mono_2}))$. As described above, it is generally accepted that the following R value

$$R = \frac{A(\delta(para))}{A(\delta(mono_1)) + A(\delta(mono_2))}$$
A: Absorbance (25)

increases with the degree of polymerization of PPP.30,71,72

The IR spectrum of the present PPP shows very weak $\delta(\text{mono}_1)$ and $\delta(\text{mono}_2)$, and the R value for the present PPP (R value = 12.7) is much larger than those for PPP's reported by Kovacic and his coworkers⁷¹ (R value = 4.6) and prepared by the Grignard reagent-mediated polycondensation of 1,4-1-Br₂Ben (R value = 1.9-3.9). This fact, together with the relatively low content of bromine, indicates that the present PPP has a relatively high degree of polymerization. The present PPP contains 2.7% of bromine. This bromine content corresponds to a molecular weight of 2960 (DP (degree of polymerization) of 38) and 5930 (DP of 76), if the polymer has bromine at one terminal unit or at both terminal units, respectively. If the present PPP has bromine at only one terminal unit, another terminal unit is regarded to be a phenyl group, which is considered to originate from the intermediate Ni(C₆H₄)-polymer species (cf. eqs 21-23) giving C₆H₅-polymer species during work up after polymerization. CP/MAS ¹³C-NMR spectrum of the present PPP (solid) shows

two peaks at 137 and 127 ppm assigned to 1,4-carbons and 2,3-carbons, respectively. No signal assignable to terminal carbon (C-X or C-H) was observable. As for poly(pyridine-2,5-diyl) (PPy) prepared analogously by using 2,5-dibromopyridine and Ni(0) complex, the polymer is soluble in formic acid and has a degree of polymerization of 20-50 (determined by light scattering method) depending on preparative conditions.^{32,102}

Various poly(arylene)s including the following polymers have been also prepare by dehalogenation polycondensation using Ni(0) complexes.

$$(26)$$

$$(Ar)_{n} : (Ar)_{n} + n \text{ Ni(2)L}_{m}$$

$$(Ar)_{n} : (Ar)_{n} : (Ar)_{n} + n \text{ NiX}_{2}L_{m}$$

$$(Ar)_{n} : (Ar)_{n} : (Ar)_{n} + n \text{ NiX}_{2}L_{m}$$

$$(Ar)_{n} : (Ar)_{n} : (Ar)_{n} + n \text{ NiX}_{2}L_{m}$$

$$(Ar)_{n} : (Ar)_{n} : (Ar)_{n} + n \text{ NiX}_{2}L_{m}$$

$$(Ar)_{n} : (Ar)_{n} : (Ar)_{n} + n \text{ NiX}_{2}L_{m}$$

$$(Ar)_{n} : (Ar)_{n} : (Ar)_{n} : (Ar)_{n} + n \text{ NiX}_{2}L_{m}$$

$$(Ar)_{n} : (Ar)_{n} :$$

2.4. Synthesis by using Pd-catalyzed coupling reactions between aryl dihalides and diethynylarenes

Table 5 summarizes results of the Pd-catalyzed polycondensation between

Run	X-Ar-X†	Catalyst	Time (h)	Yield‡ (%)	Polymer obtained
	Me	ethod IV: Diacetylenic com	pound = A	k	10 - 77-141
1	$C_6H_4Br_7$	$Pd(PPh_3)_4 + 2Cul$	2	79	Polyyne-1
2	$C_6H_4I_7$	$Pd(PPh_3)_4 + 2Cul$	2	97	Polyyne-1
3	$C_4H_2SBr_3$	$Pd(PPh_3)_4 + 2Cul$	2	84	Polyyne-2
4	C ₁₄ H ₈ Br ₂	$Pd(PPh_3)_4 + 2Cul$	2	93	Polyyne-3
5	$C_5H_3NBr_2$	$Pd(PPh_3)_4 + 2Cul$	2	95	Polyyne-4
6	$C_6H_4(COCl)_2$	$Pd(PPh_3)_4 + 2Cul$	3	64	Polyyne-5
	M	ethod V: Diacetylenic com	pound = B§		
7	$C_6H_4I_2$	Pd(PPh ₃) ₄	2	97	Polyyne-1
8	$C_4H_2SBr_2$	$Pd(OAc)_2 + 2PPh_3$	3	82	Polyyne-2
9	$C_6H_4(CH_2Cl)$	$Pd(OAc)_{2} + 2PPh_{3}$	3	72	Polyyne-6

Table 5. Pd-catalyzed polycondensation between X-Ar-X and p-C₆H₄(C \equiv CH)₂ (A) or p-C₆H₄(C \equiv CMgBr)₂ (B)

X-Ar-X and p-C₆H₄(C \equiv CH)₂ (Method IV, eq. 10) or p-C₆H₄(C \equiv CMgBr)₂ (Method V, eq. 11).³⁵

$$\begin{array}{c} p\text{-}\mathrm{C}_{6}\mathrm{H}_{4}(\mathrm{C}\!\equiv\!\mathrm{CH})_{2} \\ X\text{-}\mathrm{Ar}\text{-}\mathrm{X} + p\text{-}\mathrm{C}_{6}\mathrm{H}_{4}(\mathrm{C}\!\equiv\!\mathrm{CMgBr})_{2} & \text{polyyne } 1\text{-}6 \end{array}$$

Polyynes 1-6 are obtained in high yields. They, except for polyyne 5, show absorption in the visible region and strong fluorescence as summarized in Table 6. Marvel and his coworkers have expanded the Pd-catalyzed polymerization to the synthesis of similar polymers with the $-C \equiv C$ - and arylene units.¹⁰³

Polymer	Color	Fluorescence
Polyyne-1	yellow	bluish purple
Polyyne-2	yellow	bluish purple
Polyyne-3	red	green
Polyyne-4	light yellow	purple
Polyyne-5	white	none
Polyyne-6	yellow	purple

TABLE 6. Color and fluorescence of Polyvnes 1-6

^{*} Excess NEt_3 was added. In toluene under reflux. Amount of Pd catalyst = 0.01 mol per mol of X-Ar-X.

 $[\]dagger C_5H_3NBr_2 = 2,6$ -Dibromopyridine; $C_6H_4(CH_2Cl)_2 = p$ -Bis(chloromethyl)benzene.

[‡] Yield based on the weight of carbon in the polymer (calculated from the weight and carbon content of polymer). Yield after removing toluene extractable fraction (see Experimental).

[§] In THF under reflux. Amount of Pd catalyst = 0.01 per mol of X-Ar-X.

3. STRUCTURE

3.1. Structures in the solid and solutions

As shown in Fig. 4, PPP and related aromatic hydrocarbon π -conjugated polymers are crystalline. X-ray diffraction pattern of PTh also reveals high crystallinity of the polymer. PPP and PTh are considered to have rigidly linear rod-like structures.

In the case of PPy, it is soluble in formic acid, and its rigidly linear structure in the solution has been revealed by the large degree of depolarization ($\rho_{\nu}=0.31-0.33^{32.49}$) of polarized laser light.¹⁰⁴ On the other hand, PRpy (R = CH₃)⁶⁰ and PRTh (R = n-C₆H₁₃, n-C₈H₁₇)³³ in formic acid and CHCl₃, respectively, give a ρ_{ν} value of 0, indicating that they take random coil structures in the solutions presumably due to the steric effect of the R group. Copolymers 1 and 3 also take the random coil structure in formic acid.

The theoretical limiting value of ρ_{ν} for ideally linear molecules with strong anisotropy of polarizability ($\alpha_{\rm X} \gg \alpha_{\rm y}, \alpha_{\rm z}$) of the molecule is 1/3, and observation of the limiting value for PPy indicates that the PPy molecules take the rigidly linear structure. PQ also takes such a rigid structure in formic acid. 55 Formic

Pv = 0.33,

$$\alpha_x >> \alpha_y$$
, α_z

acid seems to mainly serve as a simple solvent for PPy, PMePy, and PQ and essentially seems not to form salts with these polymers, as judged by recovery of the original polymers by natural evaporation of formic acid from the solutions under air. Both the linear PPy and PQ and non-linear PRpy and PRTh show large specific refractive index increments ($\Delta m/\Delta c = 0.35-0.60 \, \mathrm{cm}^3 \, \mathrm{g}^{-1} \, ^{33,49}$) compared with usual non-conjugated polymers ($\Delta n/\Delta c = 0.1-0.2 \, \mathrm{cm}^3 \, \mathrm{g}^{-1}$) due to the presence of mobile π -electrons along the polymer chain.

Since PPy takes the rigidly linear structure, it shows the following interesting optical properties:

(a) Painting of polymer film (e.g. film of poly(vinyl alcohol)) with a formic acid solution of PPy and ensuing removal of formic acid affords the polymer film with the linear PPy molecule in its surface region. Elongation of the polymer film leads to the alignment of the PPy molecules in the surface region (Fig. 5a), and irradiation of polarized light to the elongated polymer film (Fig. 5b) shows strong dichroism of the film (Fig. 6). When the direction of the oscillating electric field of the polarized light coincides with the direction of the elongation ($\theta = 0$ in Fig. 5b), the film absorbs the light strongly (Fig. 6(a)). On

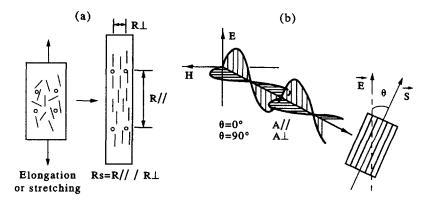


Fig. 5. (a) Stretching of polymer (e.g. poly(vinyl alcohol)) film containing the needlelike poly(pyridine-2,5-diyl) PPy molecules (the bars) in its surface region. (b) Irradiation of polarized light to the stretched film. Directions of the oscillating electric field of the polarized light and the stretching are shown by E and S, respectively.

the other hand, when the two directions are perpendicular to each other, the film absorbs the light only weakly (Fig. 6(b)). 105

(b) PPy and PBpy molecules give excimer-like emission both in formic acid with relatively high concentration (e.g. $0.5 \, \text{mol}$ of monomer units dm⁻³) and in the solid state when irradiated by UV light, whereas they show only normal monomeric fluorescence in a dilute (e.g. $2 \times 10^{-6} \, \text{mol}$ of monomer unit dm⁻³)

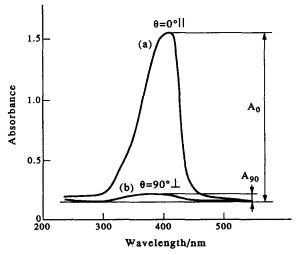


Fig. 6. Strong dichroism of the stretched poly(vinyl alcohol) film containing poly-(pyridine-2,5-diyl) (PPy) in its surface region. See Fig. 5. Measured at $\theta=0^{\circ}$ (a) and 90° (b), respectively. ¹⁰²

1178 Т. УАМАМОТО

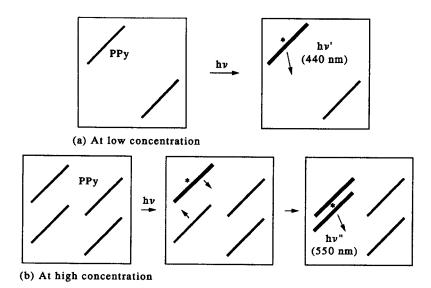


Fig. 7. Fluorescence of solutions containing a low concentration (a, e.g. 2×10^{-6} mol of monomer unit dm⁻³) and a high concentration (b, e.g. 0.5 mol of monomer unit dm⁻³) of poly(pyridine-2,5-diyl) (PPy).

formic acid solution of PPy.⁵¹ As shown in Fig. 7, a photo-excited PPy molecule with an * mark will emit usual monomeric fluorescent light ($\lambda_{max} = 440 \text{ nm}$) at the low concentration (Fig. 7a), whereas the photo-excited PPy molecule will form the excimer with the non-excited PPy molecule at the higher concentration (Fig. 7b) to show the excimer-like emission at λ_{max} of 550 nm. The linear structure of PPy seems to assist the formation of the excimer.

Because of the linear structure of PPP, PTh, PBpy, and PPy, vacuum evaporation of these polymers on substrates gives interesting oriented thin films of these polymers. Features of the vacuum evaporation and oriented structures of the thin films are as follows.

- (i) The vacuum deposition was carried out by heating the sample in a tantalum boat at $200-350^{\circ}\text{C}$ at $10^{-4}-10^{-5}$ Pa. It is reported that PPP with molecular weight of 1500-2000 corresponding to the phenylene unit of about 20 can be vaporized under vacuum. ¹⁰⁶ Since thiophene and pyridine oligomers (e.g. α -terthiophene, bipyridine, etc.) have vapor pressure analogous to that of the corresponding benzene oligomers, we assume that PTh, PPy, and PBpy with about 20 aromatic units corresponding to the length of 8–10 nm can also be vaporized.
- (ii) Figure 8 shows the electron diffraction (ED) patterns of the vacuum deposited PPP, PTh, and PBpy thin films (thickness = ca. 100 nm) on a carbon substrate. The ED patterns are taken by irradiating the electron beam perpendicularly to the surface of the thin films, and the observed ED spots can

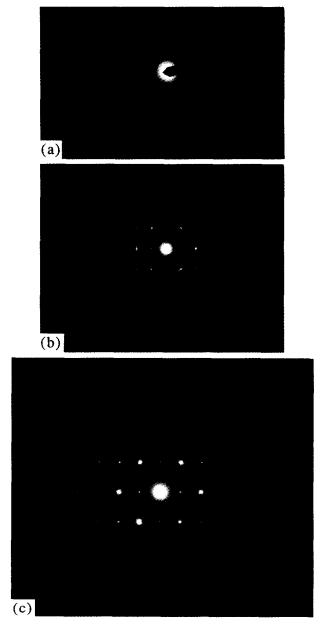


Fig. 8. Electron diffraction (ED) patterns of (a) poly(2,2'-bipyridine-5,5'-diyl) (PBpy); (b) poly(thiophene-2,5-diyl) (PTh); and (c) poly(*p*-phenylene) (PPP) deposited on carbon substrate under vacuum. 108

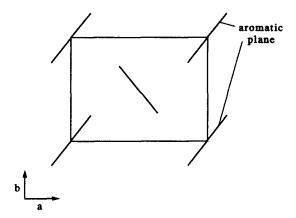


Fig. 9. Packing model for orthorhombic crystals of poly(arylene)s. The direction of the polymer chain is assigned to the c axis, and the aromatic plane of the monomer unit is shown by the bar.

be reasonably assigned by assuming the orientation of the polymer molecules perpendicularly to the surface of the substrate. The cross stripe ED spots are considered to be assigned to the diffraction concerning the *ab* plane (Fig. 9, *c* axis is the direction of the polymer chain) of orthorhombic crystals of poly-(arylene). It is reported that PPP,^{77,78} PTh,^{110,111} poly(ethylene),¹¹² and poly-(acetylene)¹¹³ take such orthorhombic or similar crystalline structures. Comparison of the values of *ab* crystalline parameters of these polymers determined by the previously reported X-ray diffractometry and the present ED patterns is shown below and is reasonable in view of the size of the monomer unit.¹⁰⁸

	PBpy	PPP	PTh	poly(ethylene)	poly(acetylene)
a(nm)	0.81	0.78	0.78	0.76	0.76
b(nm)	0.57	0.56	0.56	0.49	0.44

Molecular models for the perpendicular orientation of PPP and PTh are shown in Fig. 10.

(iii) The vacuum deposition of PPP on metal substrates also affords the oriented thin films, which show the ED patterns similar to those shown in Fig. 8. However, the number of the clearly observed ED spots varies depending on the kind of substrates, and the order of orientation judged from the number of the observable ED spots is as follows.

Order of orientation:

Number of observable diffraction spots at 150°C:

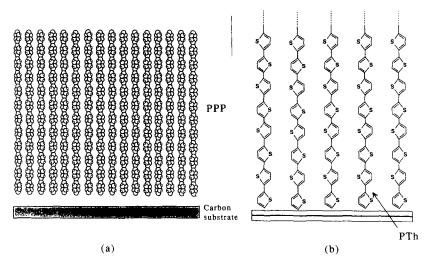


Fig. 10. Models for the perpendicular orientation of (a) poly(p-phenylene) (PPP) and (b) poly(thiophene-2,5-diyl) (PTh). Twisting of the monomer units around the bonds between the monomer units is omitted for simplification.

At r.t.:

- *Separation from spots from aluminum oxides is difficult.
- †Only dim spots were observed.

As for orientation or alignment of liquid crystalline molecules on substrate, it is generally recognized that the liquid crystalline molecules stand upright on the substrate when the surface energy or adhesive energy between the liquid crystalline molecules and the substrate is smaller than that between the liquid crystalline molecules themselves. ¹¹⁴ On the contrary, when the former energy is larger than the latter energy, the liquid molecules tend to be oriented in parallel with the surface of the substrate. The trend observed for the order of the perpendicular orientation of PPP (eq. 28) also seems to be accounted for on the basis of surface energy or strength of interaction between PPP and the substrate. It has been reported that heat of wetting $H_{\rm w}$ of carbon, aluminum, and copper in benzene, which corresponds to the monomer unit of PPP, increases in the order of

$$C(H_W = \text{ca. } 0.11 \text{ J/M}^2)^{115} < \text{Al}(H_W = \text{ca. } 0.20 \text{ J/M}^2)^{116} < \text{Cu}(H_W = 0.88 \text{ J/M}^2)^{.115}$$
 (29)

The large $H_{\rm W}$ value of Cu seems to originate from π -coordination of benzene to

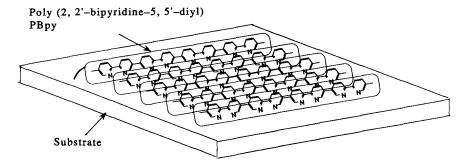
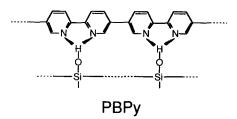


Fig. 11. Model for the parallel orientation of poly(arylene) on the surface of polar substrate.

transition metal,

and a number of π -benzene complexes of various transition metals including Cu and Ag have been isolated.¹¹⁷ Thus, the higher order of the orientation of PPP on the carbon substrate seems to be attributable to the weaker interaction of the PPP molecule with carbon than the interaction with metals.

(iv) In contrast to the vacuum deposition of poly(arylene)s to the non-polar carbon and metal substrates, similar vacuum deposition of coordinating PBpy and PTh to polar substrates like glass plate, which has acidic Si-O-H groups leads to another type of orientation of the PBpy molecule, orientation along the direction of the surface. ^{118,119} A model for the horizontal orientation is shown in Fig. 11. Coordination of the arylene unit to the polar group accounts for such a type of orientation, e.g.



2,2'-Bipyridine usually takes the *s-trans* structure. ^{120,121} However, in acidic media it takes the *s-cis* structure¹²² by interaction with the acidic proton. Because of the horizontal orientation, the vacuum deposited PBpy and PTh films are SHG (second harmonic generation)-active. ^{118,119}

Vacuum deposition of the linear PBpy and PTh on a rubbed polyimide film leads to orientation of the polymer molecules along the direction of rubbing. 108

The polyimide film containing the vacuum deposited polymer molecules shows dichroism, which is accounted for by assuming the orientation of the polymer molecules along the direction of rubbing.

3.2. Structures of doped polymers

Doping of π -conjugated polymers causes changes of the crystal structures of the polymers, ¹²³⁻¹²⁷ usually including expansion of the distance between the polymer chains and disordering of the polymer molecules.

We now describe reversible changes of chemical and crystallographical structures of PTh during iodine-doping and undoping. The present PTh prepared by the organometallic techniques has highly crystalline linear structure with well-defined linkage between the monomer units and is suited for the change of the structures.

Figures 12 and 13 show changes of IR spectrum and X-ray diffraction pattern of PTh during the iodine-doping and undoping with hydrazine, respectively.³³ The doping causes profound change in the IR spectrum even at relatively low doping level (iodine/PTh = 16–40 wt %, spectra b and c in Fig. 12) presumably due to the change in the electronic state of PTh. The iodine doping causes broadening of the X-ray diffraction peaks at the relatively low doping level, however, at 16 wt % of iodine per PTh (pattern b in Fig. 13) no new peak is observable.

At 40 wt % of iodine per PTh (pattern c in Fig. 13), a small broad rise at about $2\theta = 9^{\circ}$ (CuK α) appears in addition to the broadening of the diffraction peaks. At 69 wt % of iodine (pattern d), the peak at about $2\theta = 9^{\circ}$ due to partial formation of new structure becomes distinct, and when PTh is extensively doped (saturation doping) with iodine (120 wt %, pattern e) the X-ray diffraction pattern is entirely changed, demonstrating formation of new structure.

The four X-ray diffraction peaks observed for the extensively iodine-doped PTh at $2\theta = 8.8$, 18.0, 27.0, and 36.1° correspond to d values of 1.00, 0.49, 0.33, and 0.25 nm. The appearance of the X-ray diffraction peaks at d = a/n (a = 1.00 nm, n = 1, 2, 3, 4) indicates a one dimensional crystalline order with a period of 1.00 nm.

Based on the X-ray diffraction data as well as resonance Raman and ¹²⁹ I Mössbauer data, ¹²⁸ PTh doped to saturation with iodine is proposed to have the packing structure shown in Fig. 14. In Fig. 14, the length of the four thiophene-2,5-diyl units [Th₄] roughly fits the length of [I–I–I ··· I–I]⁻, and the 120 wt % of iodine per PTh corresponds to about 5:8 molar ratio between [I–I–I ··· I–I]⁻ and Th₄. Observed strength ratios between the four new X-ray diffraction peaks of the PTh doped to saturation (Fig. 13e) roughly agrees with calculated values based on this 5:8 molar ratio.

The IR and powder X-ray diffraction pattern (curves f in Figs 12 and 13) after the undoping of the iodine-doped PTh with hydrazine, are essentially the same

1184 T. YAMAMOTO

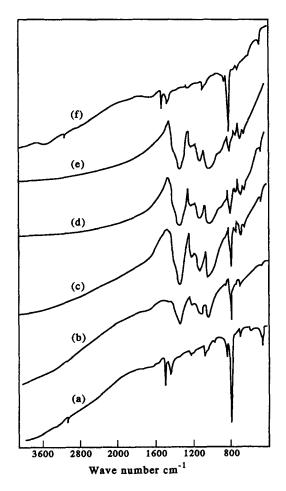


Fig. 12. Change of IR spectrum of poly(thiophene-2,5-diyl) (PTh) on doping with iodine and undoping with hydrazine. 33 (a) Original PTh; (b) iodine/PTh (wt/wt) = 0.16; (c) iodine/PTh = 0.40; (d) iodine/PTh = 0.69; (e) iodine/PTh = 1.20 (saturation doping); and (f) undoped with hydrazine.

as those of original PTh, revealing a good reversibility of the doping-undoping processes. Doping and undoping of PTh with FeCl₃ and hydrazine, respectively, are also reversible in view of the change in the IR spectrum, however, they are irreversible in view of the change in the X-ray diffraction pattern.

4. PROPERTIES

4.1. Thermal stability

As described above, PPP and other poly(arylene)s have good thermal stability

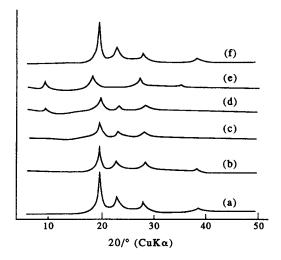


Fig. 13. Change of powder X-ray diffraction pattern of PTh on doping with iodine and undoping with hydrazine. (a)-(f): as in Fig. 12.

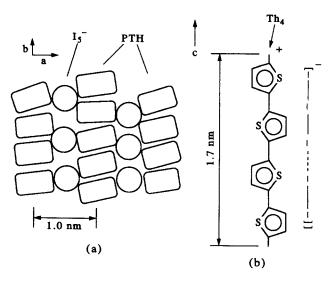


Fig. 14. Structure of PTh doped to saturation with iodine.³³

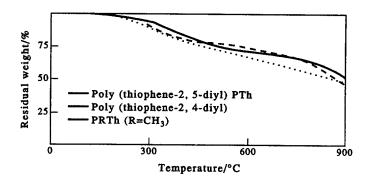


Fig. 15. Thermogravimetric analysis of poly(thiophene-2,5-diyl) (PTh), poly-(thiophene-2,4-diyl), and poly(3-methylthiophene-2,5-diyl) (PRTh) ($R = CH_3$) under N_3 .

(Table 2). Examples of TGA curves for PTh, poly(thiophene-2,4-diyl), and $Br(R = CH_3)$ under N_2 are shown in Fig. 15. Heating the fluorescent yellow PPP at 350°C causes some color change to brown.

4.2. Chemical properties

The above mentioned poly(arylene)s have high stability toward oxygen and moisture in air in contrast to high sensitivity of poly(acetylene) under air. No observable change in IR spectra has been observed for PPP and PTh stored under air for more than 15 and 13 years, respectively.

PBpy forms complexes with various metal salts and complexes,52

and some of the complexes show catalytic activity and unique electrochemical properties. In contrast to PBpy, Ppy has only a weak coordinating ability toward the metal compounds,⁵² presumably due to the head-to-tail units of Ppy which are considered to be the main components of Ppy.

$$\cdots \longrightarrow N \longrightarrow ML_m \longrightarrow minor formation of complex$$
 (31)

However, Yanagida and his coworkers reported that PPP-Ru^{131,132} and PPy-Ru¹³³ systems show good visible-light-driven photocatalytic activity, e.g.

$$H_2O \xrightarrow{h\nu, PPy-Ru} H_2$$
 (32)

Formation of an adduct of Cu(II) with PTh and CV (cyclic voltammetry) data of the PTh-Cu complex are also reported. 129,130

PQ and its analogues can be quaternized with organic halides. 134

PCyh has a structure with similarities to those of both PPP and poly(acetylene).

The color (black) of PCyh indicates the formation of an effective π -conjugation system along the polymer chain, presumably due to the coplanarity of the polymer chain in the following *s*-*cis* conformation.

In this conformation, the = C-H hydrogen of the diene unit can get between the $-CH_2$ hydrogens of the $-CH_2-CH_2$ unit, to form the planar polymer chain. In the case of PPP, coplanarity of the polymer chain is not possible due to the steric repulsion between the o-hydrogens. PCyh is air-sensitive and oxidized by oxygen in air, similarly to poly(acetylene); the high sensitivity of PCyh and poly-(acetylene) seems to be related to the effective π -conjugation of the polymers.

4.3. Electrically conducting properties

4.3.1. Chemical doping — Conversion of π -conjugated polymers into electrically

1188 T. YAMAMOTO

conducting materials by chemical doping has been widely investigated, and a number of books have been published concerning the electrically conducting π -conjugated polymers. ¹³⁵⁻¹⁴² Chemical doping by adding electron-acceptor (e.g. iodine, FeCl₃, AsF₅, etc.) or electron donor (e.g. Na, Li, etc.) to the polymer is considered to generate positive carrier (hole) or negative carrier (electron) in the π -conjugated system, and models for such carriers (soliton, polaron, bipolaron, etc.) have been given, e.g.

In the case of PTh prepared by the above described organometallic technique, reactions with I_2 and FeCl₃ produce I_5^- and a mixture of FeCl₂ and FeCl₄, respectively:

$$PTh + 2FeCl_3 \rightarrow doped-PTh^+ \cdots FeCl_4^- + FeCl_2$$
 (37)

bipolaron

as proved by Raman and Mössbauer spectroscopy. 128,143,144

The formation of I_5^- , FeCl₂ and FeCl₄⁻ supports the generation of the positive carrier in the polymer chain (*p*-type conductor) *via* electron transfer from the polymer to the electron acceptors. This *p*-type conducting property has been physically confirmed by measuring, e.g. Seebeck effect¹⁴⁵ and Hall effect. On the other hand, *n*-type conducting materials are obtained by Na doping of the π -conjugated polymers.

The electrical conductivity (σ) of PTh prepared using the Ni(0)-complex as the reactant (eq. 9) has been measured with:

- (a) a disk prepared by pressing a powder of iodine- or FeCl₃-doped PTh, and
- (b) the above described vacuum deposited film on a glass substrate.

Figure 16(a) shows the dependence of the σ -value on the amount of iodine taken up by PTh, whereas Fig. 16(b) shows the change of the σ -value with the exposure time of the vacuum deposited PTh film to the vapor of iodine.

PRTh, PORTh, PRPyrro, and other related polymers are also converted into electrically conducting materials by reaction (doping) with iodine. Copolymers containing thiophene-2,5-diyl and thiophene-2,4-diyl units

$$(s)_m (s)_n$$

show electrical conductivity when doped with iodine. However their electrical conductivity was much smaller than that of the PTh-iodine adduct.^{38,40}

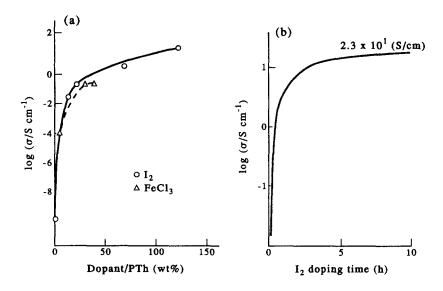


Fig. 16. (a) Dependence of the electrical conductivity (σ) of poly(thiophene-2,5-diyl) (PTh) on the amount of dopant.³³ (b) Change of the σ -value of the vacuum deposited PTh film with time of exposure to the vapor of iodine.

In contrast to PTh, PRTh, PORTh, PRPyrro, and related polymers which are constituted of electron-rich aromatic rings, ¹⁴⁶ PPy, PBpy, PPyrim, and PQ, which are constituted of electron-deficient aromatic rings, ¹⁴⁶ do not show electrically conducting properties when treated with electron acceptors like iodine. However, they are converted into electrical conductors when treated with electron donors like Na.

For example, treatment of PPy with iodine, BF_3 , or AsF_5 gives adducts with electron acceptors or Lewis acids, ^{48,49} e.g.

however, the adducts are not electrically conductive ($\sigma=1\times10^{-11}~{\rm S\,cm^{-1}}$ for the adduct with AsF₅ and 4.3 \times 10⁻⁷ S cm⁻¹ for (C₅H₃N · 0.46I)_n). This is in sharp contrast to the highly electrically conducting property of AsF₅-doped PPP. ¹³⁵⁻¹⁴² In the case of PPy, AsF₅ seems to form the adduct through the lone pair electrons of the nitrogen of PPy. This type of adduct formation seems to give only a minor effect on the 6π electron system and not to generate the carriers in the π -conjugation system. On the other hand, PPP does not have such lone pair electrons and the interaction of the Lewis acid with PPP takes place through the π -electrons, giving a profound influence for the π -conjugation system to generate the positive carrier (hole). It is generally accepted in the field

of organic chemistry that pyridine shows electron-donating Lewis base nature, originating from the lone pair electrons of nitrogen, while its 6π -conjugation system has an electron-deficient character. ¹⁴⁶

Doping of yellow PPy with sodium naphthalide gives reddish purple Na-doped PPy with an electrical conductivity of $1.2 \times 10^{-1} \, \mathrm{S \, cm^{-1}}$ as measured with pressed powder.

The electron accepting property of PPy account for this. PBpy, PPyrim, PQ, and related polymers comprised of π -deficient heterocycles show analogous doping behaviors.

PFc has electron-donating properties and forms various adducts with electron acceptors including iodine, tetracyanoquinodimethane (TCNQ), and 2,3-dichloro-5,6-dicyano-1,4-quinone, 147,148 PFc is partly oxidized by the electron accepting TCNQ to form a Fe²⁺ –Fe³⁺ mixed valent material, which shows electrical conductivity of 4.1 \times 10⁻² S cm⁻¹ (the monomer unit: TCNQ = 1:0.64). Mössbauer spectra of the PFc–TCNQ adduct reveal the presence of Fe²⁺ and Fe³⁺ species at -193° C and a rapid electron exchange between Fe²⁺ and Fe³⁺ on Mössbauer time scale (10⁻⁷ s) at room temperature; e.g.

the electrical conducting property of the PFc-TNCQ adduct may be attributed to the rapid exchange of electrons between the Fe²⁺ and Fe³⁺ species.

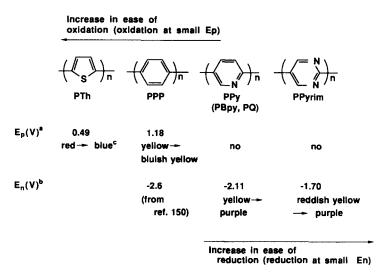
Polyyne 3 becomes an electrically conducting material ($\sigma = 4 \times 10^{-5} \, \mathrm{S \, cm^{-1}}$) when exposed to iodine vapor.³⁵

4.3.2. Electrochemical doping - PRTh, PPy, PBpy, PPyrim, PQ and its analogues, Copoly 1-3, PMePy, and several other polymers prepared by the above-

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described organometallic technique are soluble in solvents. Painting electrodes (e.g. metal, ITO, and carbon) with the polymer solution and ensuing removal of solvent affords polymer-coated electrodes suited to electrochemical analysis. The vacuum evaporation of PPP, PTh, PPy, and PBpy on metal, ITO, and carbon substrates also affords good electrodes for the electrochemical analysis of polymers (e.g. cyclic voltammetry for the polymer). The electrochemical analysis of the polymer film electrodes reveals the following electrochemical properties of the polymers.

(a) Redox potential (vs Ag/Ag⁺) of the π -conjugated polymers varies as follows, ^{54,149}



- a $E_p = 1/2$ (p-doping potential + p-undoping potential) vs. Ag/Ag⁺.
- b $E_n = 1/2$ (n-doping potential + n-undoping potential) vs. Ag/Ag^{+} .
- c Color change accompanying the doping.

If we take into account the lower E_p value of poly(pyrrole-2,5-diyl) (PPyrro)

of the above described poly(arylene)s, the ease of oxidation and reduction of poly(arylene)s appears to reflect the ease of oxidation (abstraction of electron) and reduction (addition of electron) of the arylene monomer unit, which is in the following order. 48,156,157

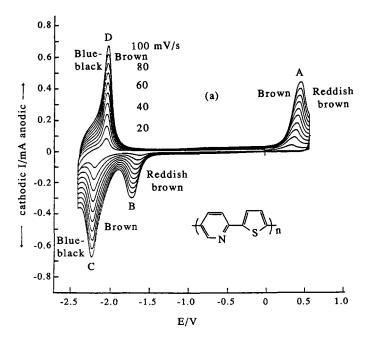
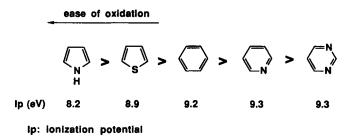


Fig. 17. Cyclic voltammogram of Copoly 1.⁵⁷ Potential: vs Ag/Ag⁺. A, p-doping; B, p-undoping; C, n-doping; D, n-undoping.



(b) In the case of copolymers consisting of electron-donating thiophene units and electron-accepting pyridine units, e.g.

their CV shows a large difference between p-doping potential and p-undoping potential as shown in Fig. 17, and the large difference is accounted for by an EC (electronic-chemical) process involving shift of a lone pair electron from the nitrogen of the pyridine ring to the p-doped thiophene ring. ⁵⁷⁻⁵⁹

No.	Poly(arylene)	Solvent	$\lambda_{\rm max}/{\rm nm}$	Ref.
1	PTh	CHCl ₃	420	39
2	(soluble part) poly(thiophene-2,4-diyl) (soluble part)	CHCl ₃	290	39
3	65:35 copolymer of thiophene-2,5-diyl and thiophene-2,4-diyl	CHCl ₃	ca. 400	39
4	PRTh $(R = n - C_6 H_{13})$	CHCl ₃	370	158
5	PPv	НСООН	373	51
6	PBpy	HCOOH	373	51
7	PMePy	HCOOH	310-340*	60
8	PPyrim	H_7SO_4	361	54
9	PO	нсоон	322, 343	55
10	Copoly 1	HCOOH	490	57
11	Copoly 2	НСООН	430	57

TABLE 7. Position of the absorption bands of poly(arylene)s prepared by organometallic methods

4.4. Optical properties

4.4.1. *UV-visible absorption* — The above described π -conjugated poly(arylene)s give rise to strong π - π * absorption bands. Examples of positions of the absorption bands are shown in Table 7.

The positions of the π - π * absorption bands of PTh (No. 1 in Table 7) and PRTh (No. 4) are comparable to those of PTh and PRTh prepared by the other methods (e.g. oxidation of thiophene and 3-alkylthiophene). A-135-142,159 Poly-(thiophene-2,4-diyl), having no extensively π -conjugated system, shows the absorption only in the UV region (No. 2). Change of the position of the absorption peak is much less sensitive (No. 3) to the change of the monomer composition of copolymers consisting of thiophene-2,5-diyl and thiophene-2,4-diyl monomer units than is change of the electrical conductivity of the copolymers with change of the monomer composition.

PPy gives rise to a relatively sharp absorption peak at about 370 nm, and the λ_{max} value is comparable to that (ca. 380 nm measured in the solid state) of PPP prepared by the organometallic method. The position of λ_{max} of PPy is virtually independent of the degree of polymerization (DP) in a DP range of 16-50.

^{*} Depending on the position of the Me group.

$$\lambda_{\text{max}} = 372 \pm 3 \quad \text{for } n = 16-50$$
PPv

The introduction of the alkyl group R to PTh and PPy causes a shift of the π - π * absorption band to lower wavelength, which implies lower degree of π -conjugation in the R-substituted poly(arylene)s.

It may be expected that Copoly 1 gives rise to a UV-visible absorption peak at a middle position between the absorption peaks of PTh and PPy. However, the absorption band of Copoly 1 with λ_{max} at 490 nm (No. 9 in Table 7) rather shows a red shift compared with those of PTh and PPy, presumably due to intramolecular charge transfer.

$$-\sqrt{\frac{\delta_{-}}{s}}$$

4.4.2. Fluorescence, electroluminescence, and non-linear optical properties — The π -conjugated poly(arylene)s and polyynes described above generally show strong fluorescence and several data are given in Table 6. Linear poly(arylene)s like PPy give rise to excimer-like emission as shown in Fig. 7.

Fluorescent (photoluminescent) materials are expected to have electroluminescent properties. Actually PTh¹⁶⁰ and PRTh¹⁶¹ films do show electroluminescent properties. An Al/PTh/ITO electric junction prepared with PTh prepared by the organometallic method gives a i-V (current-voltage) curve which is accounted for by assuming the presence of a Schottky barrier between Al and PTh (vide infra), and the electric junction emits visible light centered at about 550 nm when direct current voltage of 25 V is applied. 149,160

As described above, PBpy and PTh molecules arranged along the direction of the surface of the glass substrate (Fig. 11) are SHG (second harmonic generation) active. On the other hand some of the π -conjugated poly(arylene)s show another type of non-linear optical property, THG (third harmonic generation).

Among the poly(arylene)s, copolymers consisting of electron-donating thiophene rings and electron-accepting pyridine rings e.g.

show large THG with $\chi^{(3)}$ value of

$$\chi^{(3)} = 1.5 \times 10^{-10} \text{ esu}$$
 (42)

at a wavelength of 1.525 μ m (3 ω). Figure 18 shows the dependence of the $\chi^{(3)}$

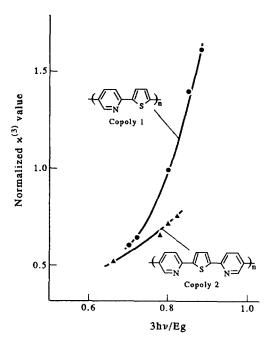


Fig. 18. χ^3 vs hy normalized by optical band-gap energy for Copoly 1 and Copoly 2. 162

value on wavelength normalized by the optical band-gap energy. Since Copoly 1 gives a mechanically strong thin film with smooth surface, it is suited to the making of optical devices based on THG.

5. APPLICATION

5.1. Battery

The poly(arylene)s prepared by organometallic methods serve as active materials and electrodes of galvanic cells (primary cells and secondary cells (batteries)). Usability of PTh, ¹⁶³ poly(acetylene), ¹⁶⁴ and PPP¹⁶⁵ as active materials and electrodes of polymer batteries was first reported at almost the same time (1981–1982).

For example, the following charging reactions (eqs 43-47) take place when a ZnI_2 solution is electrolyzed by using Zn and PTh as the negative and positive electrodes, respectively.

1196 T. YAMAMOTO

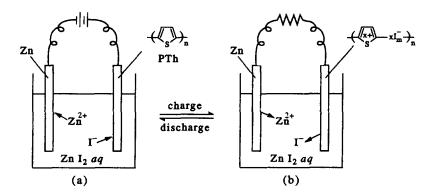


Fig. 19. Charging and discharging of Zn|ZnI₂|PTh battery. The state (a) is charged to the state (b), which can go back into the state (a) by discharging.

Negative electrode:
$$Zn^{2+} + 2e \longrightarrow Zn \qquad (43)$$
Positive electrode:
$$2l^{-} \longrightarrow l_{2} + 2e \qquad (44)$$

$$l_{2} + (-) \longrightarrow lodine \qquad lodine doped-PTh \qquad (45)$$
PTh (45)

$$Znl_2 + PTh \xrightarrow{charge} Zn + iodine doped-PTh$$
 (47)

Since the iodine doped-PTh has electrically conducting properties and stores positive charge in the polymer chain, it serves as a good polymer electrode for a battery (Fig. 19). When the battery is discharged, a reverse reaction (cf. eq. 47) takes place. An example of the charge-discharge profile of the Zn|ZnI₂| PTh battery is shown in Fig. 20. ^{166,167} As shown in Fig. 20, the charging and discharging proceeds smoothly with high current efficiency and energy efficiency. ^{166,167}

The charging reaction (eq. 46) at the positive electode is regarded as the electrochemical p-doping (or anion doping) of PTh, and such a battery can be expanded to various types of batteries using various metals, electrolytes, and polymer electrodes. Among the batteries, lithium batteries are especially important since they show high discharging voltage (ca. 3V).

$$Li^+ + e \xrightarrow{\text{charge}} Li$$
 (48)

$$Y^- + polymer \xrightarrow{charge} p$$
-doped polymer + e (49)

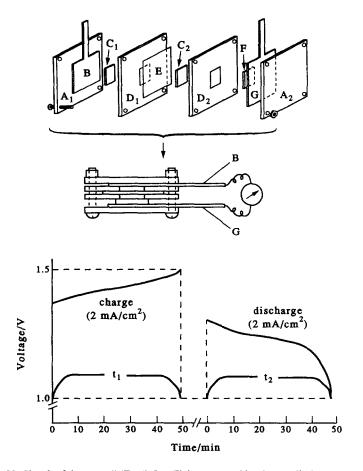


Fig. 20. Sketch of the test cell (Zn|ZnI₂|PTh battery) and its charge–discharge curves (lower figure). 166 A₁ and A₂, board; B, negative electrode (Zn); C₁ and C₂, glass fiber (Toyo Roshi type GA 100, $10\times10\,\text{mm}$) impregnated with an aqueous solution (0.1 cm³) of ZnI₂ (0.5 mol/dm³); D₁ and D₂, board with hole (10 \times 10 mm); E, cation-exchange membrane; F, polymer-coated carbon fiber plate (10 \times 10 mm); G, Pt-plate.

LiY + polymer
$$\stackrel{\text{charge}}{\rightleftharpoons}$$
 Li + p-doped polymer (50)
Y = ClO_4^- , BF₄, etc.

The poly(arylene)s prepared by organometallicmethods as well as electrochemical processes are usable as positive electrodes of lithium batteries. 168

On the other hand, the iodine-poly(arylene) adducts are useful as active materials of lithium-iodine solid electrolyte cells. Since the lithium-iodine solid electrolyte cell uses no solvent, one need not worry about the leakage of electrolytic solution from the cell, and the cell is suited for long-term use

(e.g. cells for pacemakers).

$$2Li \rightarrow 2Li^{+} + 2e \tag{51}$$

$$I_2 + 2e \rightarrow 2I^- \tag{52}$$

$$2Li + I_2 \xrightarrow{\text{solid electrolyte}} 2Lil$$
 (53)

Iodine in the lithium-iodine solid electrolyte cell, however, is not used as pure iodine due to its very low electrical conductivity; iodine is used as an electrically conducting charge-transfer complex with an electron-donating polymer (e.g. poly(vinylpyridine)). ¹⁶⁹

Since the iodine adducts of the poly(arylene)s have much higher electrical conductivity than the iodine-poly(vinylpyridine) adduct, the iodine adducts of the poly(arylene)s are expected to serve as better active materials for the lithium-iodine solid electrolyte cells. Actually use of poly(arylene)s prepared by organometallic methods (e.g. PTh) as a partner for iodine affords an excellent lithium-iodine solid electrolyte cell. Structure of the cell and its discharging profile is shown in Fig. 21. ¹⁷⁰ As shown in Fig. 21, the cell maintains a discharging voltage higher than 2.3 V up to 85.2% utilization of iodine.

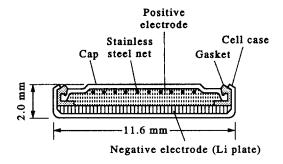
5.2. Electrochromic display

Doping of poly(arylene)s by chemical methods (e.g. with concentrated $H_2SO_4^{40}$) as well as electrochemical methods leads to a deep change of their color, which is interpreted as due to formation of polarons and/or bipolarons, ^{135–140} e.g.

In the case of PORTh, a shift of a lone pair electron of the OR group seems to occur in the p-doped state.⁸⁰

Since the electrochemical oxidation (eq. 54) is reversible, preparation of electrochromic displays becomes possible by using:

(i) ITO electrode coated with a poly(arylene) film prepared by a painting method (painting the ITO electrode with a solution containing poly(arylene) (e.g. PRTh)), and



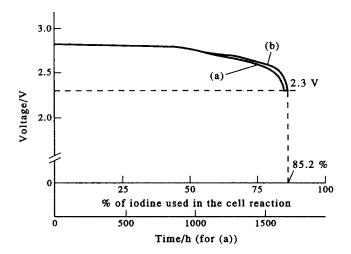


Fig. 21. Structure of the Li|LiI| I_2 -PTh adduct cell and discharging profile of the cell. Positive electrode = I_2 -PTh adduct. (a) Wt-% of iodine = 75% (weight of the adduct = 61.2 mg); (b) Wt-% of iodine = 50%. ¹⁷⁰

(ii) ITO electrode coated with a vacuum deposited film of poly(arylene) (e.g. PTh).

Examples of the color change accompanying the electrochemical redox reactions are shown in Section 4.3.2.

5.3. Others

The poly(arylene)s prepared or molecularly designed based on organometallic chemistry are now going to be used, or are intended to be used in the following various areas:

(a) Conductors or materials to remove static electricity. Various fibers, films, ¹⁷¹ and related materials coated with the conductive polymers are now industrialized. The polymers are to be used in electron beam lithography to

avoid charge accumulation due to electrons from electron beam.¹⁷² Recently industrial use of the polymers as electrodes of aluminum capacitor¹⁷³ as well as base layers for plating metals through holes in circuit boards has been established.

- (b) Electrode and electochromism. The doping-undoping or redox property of the conductivy polymers has been utilized to store energy in batteries and to make electrochromic displays (ECD). Examples of the storage of energy in poly(arylene)s and electrochromism of poly(arylene)s are described above. Photo-assisted reduction of water and ketones has been reported. Poly-(iso-thianaphthene) is expected to be usable as a transparent conducting material.
- (c) Electric devices. Electric devices including diodes, $^{149,177-179}$ transistors, $^{180-182}$ and solar cells 183,184 have been made by using the π -conjugated conductive polymers. Among the polymers, poly(thiophene-2,5-diyl) and its derivatives are most widely used.
- (d) Optical devices. As described above, π-conjugated poly(arylene)s show photoluminescent, electroluminescent and non-linear optical properties. Because of these interesting properties, the polymers are expected to support future industries producing highly effective optical telecommunication systems, photocomputer systems, photoconducting devices, and other optical devices.¹⁴¹
- (e) Other. Since poly(arylene)s possess interesting π -conjugated electron systems along the polymer chain, appropriate molecular design of poly-(arylene)s is expected to produce interesting materials, e.g. organic superconductors, organic ferromagnetic materials, molecular devices, etc.

6. CONCLUSION AND SCOPE

The basic C-C coupling reactions using transition metal complexes (e.g. eq. 1) found by us provide new polymerization methods (Methods I-V), which make it possible to make poly(arylene)s and poly(ethyne-diyl-arylene)s, $(Ar)_{\overline{n}}$ and (C≡C-Ar)_n. The new polymerization methods have been especially developed to design electrically conducting and stable (chemically and thermally) poly(arylene)s, and various such poly(arylene)s including poly(thiophene-2,5diyl), the first conductive polymer consisting of 5-membered heteroaromatic rings, poly(pyridine-2,5-diyl), and poly(3-alkylthiophene-2,5-diyl), the first conductive polymer having substituents, were first prepared by these techniques. The poly(arylene)s have well-defined bonding between the monomer units and are suited to reveal chemical (e.g. redox properties and coordinating properties), electrical and optical properties as well as chemical and physical (crystallographic) structures of poly(arylene)s. Some of the poly(arylene)s thus prepared have linear and rigid structures, and alignment of the polymers on the surface of substrate as well as in the surface region of polymer (e.g. poly(vinyl alcohol)) films is possible. 102,185 The poly(arylene)s are useful materials for

making devices such as polymer batteries, electrochromic devices, electroluminescent devices, photoluminescent devices, etc. They are also industrially used as materials to remove static electricity and as electrodes. Some of the devices were first developed by us using the poly(arylene)s.

Thus, the basic pure organometallic reactions (e.g. eq. 1) have provided a basis to design electrically conducting and stable polymers, which show interesting electrical and optical properties. Such polymers are expected to support future industries.

REFERENCES

- T. YAMAMOTO, A. YAMAMOTO and S. IKEDA, J. Am. Chem. Soc. 93, 3350, 3360 (1971).
- 2. T. KOHARA, T. YAMAMOTO and A. YAMAMOTO, J. Organomet. Chem. 192, 265 (1980).
- 3. K. Tatsmi, A. Nakamura, S. Komiya, T. Yamamoto and A. Yamamoto, *J. Am. Chem. Soc.* **106**, 8181 (1984).
- 4. S. KOMIYA, Y. ABE, A. YAMAMOTO and T. YAMAMOTO, Organometallics 2, 1466 (1983).
- P. W. JOLLY and G. WILKE, The Organic Chemistry of Nickel, Vol. I, p. 206, Academic Press, New York (1974).
- 6. J. P. COLLMAN, L. S. HEGEDUS, J. R. NORTON and R. G. FINKE, Principles and Applications of Organotransition Metal Chemistry, p. 322, University Science Book, Mill Valley (1987).
- 7. J. K. KOCHI, Organometallic Mechanisms and Catalysis, Academic Press, New York (1978).
- 8. K. TAMAO and M. KUMADA, In the Chemistry of the Metal-Carbon Bond, (F. R. HARTLEY Ed.), Vol. 4, p. 819, Interscience, New York.
- 9. A. YAMAMOTO, Organotransition Metal Chemistry, p. 244, John Wiley, New York (1986).
- 10. A. J. Pearson, Metalloorganic Chemistry, p. 36, John Wiley, New York (1985).
- 11. B. AKERMARK and A. LJUNGQVIST, J. Organomet. Chem. 149, 97 (1978).
- B. AKERMARK, H. JOHNASEN, B. ROSS and U. WAHLGREN, J. Am. Chem. Soc. 101, 5876 (1979).
- 13. P. BINGER and M. J. DOYLE, J. Organomet. Chem. 162, 195 (1978).
- K. Tamao, K. Sumitani, Y. Kiso, M. Zembayashi, A. Fujioka, K. Kodama, I. Nakajima. A. Minato and M. Kumada, Bull. Chem. Soc. Jpn 49, 1958 (1976).
- 15. R. J. P. CORRIU and J. P. MASSE, J. Chem. Soc., Chem. Commun. 144 (1972).
- 16. D. G. MORRELL and J. K. KOCHI, J. Am. Chem. Soc. 97, 7262 (1979).
- 17. A. S. KENDE, L. S. LIEBESKIND and D. BRAITSCH, Tetrahedron Lett. 3375 (1977).
- 18. M. IYODA, M. SAKITANI, H. OTSUKA and M. ODA, Chem. Lett. 127 (1985).
- 19. M. ZEMBAYASHI, K. TAMAO, J. YOSHIDA and M. KUMADA, Tetrahedron Lett. 4089 (1977).
- 20. M. F. SEMMELHACK, P. M. HELQUIST and L. D. JONES, J. Am. Chem. Soc. 93, 5903 (1971).
- 21. M. F. SEMMELHACK and L. S. RYONO, J. Am. Chem. Soc. 97, 3873 (1975).
- 22. A. S. KENDE, L. S. KIEBESKING and D. M. BRAITSCH, Tetrahedron Lett. 3375 (1975).
- M. TIECCO, L. TESTAFERRI, M. TINGOLI, D. CHIANELLI and M. MONTUCCI, Synthesis 736 (1984).
- 24. Z.-H. ZHOU and T. YAMAMOTO, J. Organomet. Chem. 414, 119 (1991).
- 25. T. YAMAMOTO, S. WAKABAYASHI and K. OSAKADA, J. Organomet. Chem., 428, 223 (1992).
- 26. Y. THODA, K. SONOGASHIRA and N. HAGIHARA, Synthesis 777 (1977).
- 27. K. SONOGASHIRA, Y. THODA and N. NIGHARA, Tetrahedron Lett. 4467 (1975).
- 28. A. SEKIYA and N. ISHIKAWA, J. Organomet. Chem. 118, 349 (1976).
- 29. T. YAMAMOTO and A. YAMAMOTO, Chem. Lett. 353 (1977).
- 30. T. YAMAMOTO, Y. HAYASHIA and A. YAMAMOTO, Bull. Chem. Soc. Jpn 51, 2091 (1978).
- 31. T. YAMAMOTO, K. OSAKADA, T. WAKABAYASHI and A. YAMAMOTO, Makromolek. Chem. Rapid. Commun. 6, 671 (1985) Jpn. Kokai Pat. 1986–233014.
- 32. Т. Уамамото, Т. Iто and K. Kubota, Chem. Lett. 153 (1988) Jpn. Kokai Pat. 1989–178517 (1989).
- 33. T. Yamamoto, A. Morita, Y. Miyazaki, T. Maruyama, H. Wakayama, Z.-H. Zhou, Y. Nakamura, T. Kanbara, S. Sasaki and K. Kubota, *Macromolecules* 25, 1214 (1992).

- T. Yamamoto, A. Morita, T. Maruyama, Z.-H. Zhou, T. Kanbara and K. Sanechika, *Polym. J.* 22, 187 (1990) Jpn. Kokai Pat. 1990–8213 (1990); K. Sanechika, T. Yamamoto and A. Yamamoto, *Bull. Chem. Soc. Jpn* 57, 752 (1984).
- 35. K. SANECHIKA, T. YAMAMOTO and A. YAMAMOTO, Bull. Chem. Soc. Jpn 57, 753 (1984).
- 36. K. SANECHIKA, T. YAMAMOTO and A. YAMAMOTO, Polym. Prepr. Jpn 28, 966 (1979).
- 37. Т. YAMAMOTO, K. SANECHIKA and A. YAMOMOTO, J. Phys. Sci., Polym. Lett. 18, 9 (1980).
- 38. K. SANECHIKA, T. YAMAMOTO and A. YAMAMOTO, J. Polym. Sci., Polym. Lett. 20, 365 (1982).
- 39. T. YAMAMOTO, K. SANECHIKA and A. YAMAMOTO, Bull. Chem. Soc. Jpn 56, 1497 (1983).
- 40. Т. YAMAMOTO, K. SANECHIKA and A. YAMAMOTO, Bull. Chem. Soc. Jpn 56, 1503 (1983).
- 41. Т. YAMAMOTO, K. SANECHIKA and A. YAMAMOTO, Japan Kokai Patent 1981–47421 (1981) & 1983–96626 (1983) (Jpn. Pat., 1200926 (1984) & 1236581 (1984)).
- 42. T. YAMAMOTO and K. SANCHIKA, Chem. Ind. (London), 301 (1982).
- T. YAMAMOTO, K. SANEHIKA and A. YAMAMOTO, Japan Kokai Patent 1983–147426 (1983) & 1984–90301 (Jpn. Pat., 1258016 (1985) and 1262404 (1985)).
- 44. T. YAMAMOTO, K. SANECHIKA and A. YAMAMOTO, U.S. Pat. 4,521,589 (1985).
- 45. I. KHOURY, P. KOVACIS and H. M. GILOW, J. Polym. Sci., Polym. Lett. 19, 395 (1981).
- 46. T. YAMAMOTO and H. SAKAI, J. Mater. Sci. 4, 916 (1985).
- 47. T. YAMAMOTO, K. SANECHIKA and H. SAKAI, J. Macromolec. Sci. -Chem. A27, 1147 (1990).
- 48. T. YAMAMOTO, T. ITO, K. SANECHIKA and M. HISHINUMA, Synth. Metals 25, 103 (1988).
- 49. T. YAMAMOTO, T. ITO, K. SANECHIKA and M. HISHINUMA, Chem. Ind. (Lond.), 337 (1988).
- 50. Т. Yамамото, Japan Kokai Patent, 1989-178517 (1989).
- 51. T. YAMAMOTO, T. MARUYAMA, T. IKEDA and M. SISIDO, J. Chem. Soc., Chem. Commun. 1306 (1990).
- 52. .T YAMAMOTO, Z.-H. ZHOU, T. MARUYAMA and T. KANBARA, Chem. Lett. 223 (1990).
- 53. T. YAMAMOTO, Japan Kokai Paten, 1989-305739 (1989).
- T. KUSHIDA, T. KANBARA, T. YAMAMOTO and I. KUWAJIMA, Polym. Prepr. Jpn 40, 2284 (1981); T. KANBARA, T. KUSHIDA, N. SAITO, I. KUWAJIMA, K. KUBOTA and T. YAMOTO, Chem. Lett. (1992).
- 55. T. KANBARA, N. SAITO and T. YAMAMOTO, Macromolecules 24 5883 (1991).
- H. SAITO, K. OSAKADA and T. YAMAMOTO, *Polym. Prepr. Jpn* 40, 2281 (1991); T. YAMAMOTO,
 H. SAITO and K. OSAKADA *Polym. Bull.*, in press (1992).
- 57. Z.-H. ZHOU, T. MARUYAMA, T. KANBARA, T. IKEDA, K. ICHIMURA, T. YAMAMOTO and K. TOKUDA, J. Chem. Soc., Chem. Commun. 1210 (1991).
- 58. T. YAMAMOTO, Z.-H. ZHOU, T. MARUYAMA, T. KANBARA, T. IKEDA, K. ICHIMURA, K. TOKUDA and K. KUBOTA, to be submitted.
- M. SHIMURA, T. KANBARA, K. OSAKADA and T. YAMAMOTO, Proc. 61st Spring Annual Meet. of Chem. Soc. Jpn, p. 1392 (#4c232) (1991); T. YAMAMOTO, M. SHIMURA and K. OSAKADA, Chem Lett., 1003 (1992).
- T. MARUYAMA, Z.-H. ZHOU, T. YAMAMOTO and K. KUBOTA, Proc. 61st Spring Annual Meet. of Chem. Soc. Jpn p. 1392 (#4c231) (1991); idem., Chem. Lett., 643 (1992).
- K. SANECHIKA, T. YAMAMOTO and A. YAMAMOTO, Polym. Prepr. Jpn 30, 3B27 (1981); Bull. Chem. Soc. Jpn 57, 753 (1984).
- K. K. KANAZAWA, A. F. DIAZ, R. H. GEISS, W. D. GILL, J. F. KWAK, J. A. LOGAN, J. F. RABOLT and G. B. STREET, J. Chem. Soc., Chem. Commun. 854 (1979).
- 63. T. YAMAMOTO, A. TANIGUCHI, K. SANECHIKA and A. YAMAMOTO, *Macromolecules* 16, 1555 (1983).
- 64. R. S. ELSENBAUMER, K. Y. JEN and R. OBOODI, Synth. Metals 15, 169 (1986).
- A. CZERWINSKI, D. D. CUNNINGHAM, A. AMER, J. R. SCHRADER, C. V. PHAM, H. ZIMMER and H. B. MARK, JR, J. Electrochem. Soc. 134, 1158 (1987).
- J.-E. OSTERHOLM, J. LAASKI, P. NYHOLM, H. ISOTALO, H. STUBB, O. INGANAS and W. R. SALANECK, Synth. Metals 28, C435 (1989).
- E. E. HAVINGA, L. W. VAN HORSSEN, W. HOEVE, H. WYNBERG and E. W. MEIJIER, *Polym. Bull.* 18, 277 (1987).
- 68. M. R. BRYCE, A. CHISSEL, P. KATHIRGAMANATHAN, D. PARKAR and N. R. M. SMITH, J. Chem. Soc., Chem. Commun. 466 (1987).
- 69. J. Ruhe, T. Ezquerra and G. Wegner, Makromolek. Chem. Rapid Commun. 10, 103 (1989).
- 70. K. KAERIYAMA and H. MATSUDA, Br. Polym. J. 23, 141 (1990).

- 71. P. KOVACIC and J. OZIOMEK, J. Org. Chem. 29, 100 (1964).
- 72. P. KOVACIC and L. C. HSU, J. Polym. Sci. A-1 4, 5 (1966).
- 73. G. TOURILLON and F. GARNIER, J. Electronanal. Chem. 135, 173 (1982).
- G. TOURILLON, Handbook of Conducting Polymers (T. A. SKOTHEIM Ed.), Vol. 1, p. 293. Marcel Dekker, New York (1986).
- 75. J. E. WHANG, J. H. HAN, H. S. NALWA, T. WATANABE and S. MIYATE, *Synth. Metals* **43**, 3043 (1991).
- Z. Mo, K.-B. LEE, Y. B. MOON, M. KOBAYASHI, A. J. HEEGER and F. WUDL, *Macromolecules* 18, 1972 (1985).
- 77. S. BRUCKNER and W. PORZIO, Makromolek. Chem. 189, 961 (1988).
- 78. P. KOVACIC, M. B. FELDMAN, J. P. KOVACIC and J. B. LANDO, J. Appl. Polym. Sci. 12, 1735 (1968); G. FROYER, F. MAURICE, J. P. MERSICR, D. RIVIERE, M. LE CUN and P. AUVRAY. Polymer 7, 139 (1986).
- 79. T. YAMAMOTO, Y. EHARA and A. YAMAMOTO, Nippon Kagaku Kaishi, 310 (1982).
- 80. T. YAMAMOTO, A. KASHIWAZAKI and K. KATO, Makromolek. Chem. 190, 1649 (1989).
- 81. M. UEDA and F. ICHIKAWA, Macromolecules 23, 926 (1990).
- 82. M. UEDA, Y. MIYAJI, T. ITO, Y. OBE and T. SONE, Macromolecules 24, 2694 (1991).
- 83. S. HOTTA, M. SOGA and N. SONODA, Synth. Metals 26, 267 (1988).
- 84. K. Yoshino, S. Hayasi and R. Sugimoto, Jpn. J. Appl. Phys., 23, L899 (1984).
- 85. R. SUGIMOTO, S. TAKEDA and K. YOSHINO, Chem. Express 1 635 (1986).
- J. RONCALI, R. GARREAU, A. YASSAR, P. MARQUE, F. GARNIER and M. LEMAIRE, J. Phys. Chem. 91, 6706 (1987).
- 87. M. SATO, S. TANAKA and K. KAERIYAMA, Synth. Metals 14, 275 (1986).
- 88. M. SATO, S. TANAKA and K. KAERIYAMA, Makromolek. Chem. 188, 1763 (1987).
- 89. M. R. BRYCE, A. CHISSEL, P. KATHIRGAMANATHAN, D. PAKER and N. R. M. SMITH, J. Chem. Soc., Chem. Commun. 466 (1987).
- 90. R. M. S. Maior, K. Hinkelmann, H. Eckert and F. Wudl, *Macromolecules* 23, 1268 (1990).
- 91. M. HIDAI, T. KASHIWAHI, T. IKEUCHI and Y. UCHIDA, J. Organomet. Chem. 30, 279 (1971).
- T. YAMAMOTO, T. KOHARA and A. YAMAMOTO, Bull. Chem. Soc. Jpn 54, 1720, 2010, 2161 (1981).
- 93. G. W. PARSHALL, J. Am. Chem. Soc. 96, 2360 (1974).
- 94. D. G. MARRELL and J. K. KOCHI, ibid. 97, 8262 (1975).
- S. KOMITA, T. ALBRIGHT, R. HOFFMANN and J. K. KOCHI, ibid. 98, 7255 (1976) and 99, 8840 (1977).
- F. OZAWA, K. KURIHARA, T. YAMAMOTO and A. YAMAMOTO, Bull. Chem. Soc. Jpn 58,399 (1985).
- 97. J. CHATT and B. L. SHAW, *J. Chem. Soc.* 1719 (1960). Preliminary X-ray structural analysis of Complex 1a revealed shorter Ni-C bond length (187 pm) than the usual Ni-C bond length, supporting the partial formation of the double bonding. The large *p*-system of the anthryl group may facilitate the Ni-C double bonding. (M. R. CHURCHILL, K. L. KALRA and M. V. VEIDIS, *Inorg. Chem.* 12, 1656 (1983) and Ref. 5).
- 98. H. SHIRAKAWA and S. IKEDA, Polym. J. 2, 231 (1971).
- 99. G. BONTEMPELLI, F. MAGNO, B. CORAIN and G. SCHIAVON, J. Electroanal. Chem. 103, 243 (1979).
- 100. G. Bontempelli, F. Mango, S. Daniele and G. Schiavon, ibid. 159, 117 (1983).
- 101. M. Mori, Y. Hashimoto and Y. Ban, Tetrahedron Lett. 21, 631 (1980).
- 102. T. YAMAMOTO, T. MARUYAMA and K. KUBOTA, Chem. Lett., 1951 (1989); 1827 (1992).
- 103. D. L. TRUMBO and C. S. MARVEL, J. Polym. Sci., Polym. Chem. 24, 2311 (1986).
- 104. K. KUBOTA and B. CHU, Biopolymers 22, 1461 (1983).
- 105. T. YAMAMOTI, T. MARUYAMA and K. KUBOTA, Chem. Lett. 1951 (1989).
- C. E. BROWN, P. KOVACIC, C. A. WOLKIE, R. B. CODY, JR and J. A. KINSINGER, J. Polym. Sci., Polym. Lett. Ed. 23, 453 (1985).
- 107. T. YAMAMOTO, T. KANBARA and C. MORI, Synth. Metals 38, 399 (1990).
- T. YAMAMOTO, C. MORI, H. WAKAYAMA, Z.-H. ZHOU, T. MARUYAMA, T. KANBARA and R. OHKI, Chem. Lett. 1483 (1991).
- 109. T. Kanbara, C. Mori, H. Wakayama, T. Fukuda, Z.-H. Zhou, T. Maruyama, K.

- OSAKADA and T. YAMAMOTO, J. Chem. Phys. 83, 771 (1992).
- Z. Mo, K.-B. LEE, Y. B. MOON, M. KOBAYASHI, A. HEEGER and F. WUDL, *Macromolecules* 18 1972 (1985).
- 111. S. BRUCKNER and W. POZZIO, Makromolek. Chem. 189, 961 (1988).
- 112. E. R. WALTER and F. P. REDING, J. Polym. Sci. 21, 561 (1956).
- 113. R. H. BAUGHMAN, S. L. HSU and A. J. SIGNORELLI, Molec. Cryst. Liq. Cryst. 52, 251 (1979).
- J. COGNARD, Molecular Crystals and Liquid Crystals. Supplement Series, Supplement 1, Alignment of Nematic Liquid Crystals and Their Mixtures, pp. 43, 49, Gordon and Breach Science Publishers, New York (1982).
- 115. F. E. BARTELL and N. JACKMAN, J. Phys. Chem. 58, 36 (1954).
- 116. H. COCKRANE and R. RUDHAM, Trans. Faraday Soc. 61, 2246 (1965).
- 117. W. E. SILVERTHORN, Adv. Organomet. Chem. 13, 48, 120 (1975).
- T. FUKUDA, T. KANBARA, T. YAMAMOTO, T. FUJIOKA, K. KAJIKAWA, H. TAKEZOE and A. FUKUDA, Mol. Cryst. Liq. Cryst., accepted; Spring Annual Meet. of Soc. Appl. Jpn (1992).
- 119. T. KURATA, A. TSUMURA, H. FUCHIGAMI, H. KOEZUKA and T. YAMAMOTO, Spring Annual Meet. of Soc. Appl. Phys. Jpn (1992).
- 120. L. L. MERITT, JR and E. D. SCHROEDER, Acta Cryst. 9, 801 (1956).
- 121. F. BERTINOTTI, A. M. LIQUORI and R. PIRISI, Gazz. Chim. Ital. 86, 893 (1956).
- 122. R. H. LINNELL and A. KACZMAREZYK, J. Phys. Chem. 65, 1196 (1961).
- 123. G. PEROGO, G. LAUGLI, U. PEDRETTI and ALLEGRA, Makromolek. Chem. 189, 2687 (1988).
- R. BAUGHMAN, N. S. MURTHY, G. MILLER and L. W. SHACKLETTE, J. Chem. Phys. 79, 1066 (1983).
- 125. J. C. W. CHIEN, F. E. KARASZ and K. SHIMAMURA, Macromolecules 15, 1012 (1982).
- 126. D. R. RUEDA, M. E. CAGIAO and F. J. MELTA CALLEJA, Polym. Bull. 21, 635 (1989).
- 127. H. W. HASSLIN and C. RIEKEL, Synth. Metals 5, 37 (1982).
- H. SAKAI, M. MIZOTA, Y. MAEDA, T. YAMAMOTO and A. YAMAMOTO, Bull. Chem. Soc. Jpn 58, 926 (1985).
- 129. A. CZERWINSKI, C. V. PHAM, D. D. CUNNINGHAM, L. LAGUREN-DAVIDSON, H. ZIMMER and H. B. GRAY JR, J. Electrochem. Soc. 133, 576 (1986).
- A. CZERWINSKI, H. ZIMMER, C. V. PHAN, A. AMER, J. SCHRADER and H. B. MARK JR, Anal. Lett. 18, 673 (1985); A. CZERWINSKI, D. D. CUNNINGHAM, A. AMER, J. R. SCHRADER, C. V. PHAM, H. ZIMMER, H. B. MARK JR and S. PONS, J. Electrochem. Soc. 134, 1158 (1987).
- 131. S. YANAGIDA, A. KABUMOTO, K. MIZUMOTO, C. PAK and K. YOSHINO, J. Chem. Soc.. Chem. Commun. 474 (1985).
- 132. T. SHIBATA, A. KABUMOTO, T. SHIRAGAMI, O. ISHITANI, C. PAK and S. YANAGIDA, *J. Phys. Chem.* **94**, 2068 (1990).
- 133. S. MATSUOKA, T. KOHZUKI, A. NAKAMURA, C. PAK and S. YANAGIDA, J. Chem. Soc., Chem. Commun. 580 (1991).
- 134. T. KANBARA and T. YAMAMOTO, 61st Annual Meet. of Chem. Soc. of Jpn, Osaka, April (1992) Macromolecules, submitted.
- T. A. SKOTHEIM (Ed.), Handbook of Conducting Polymers, Vols I and II, Marcel Dekker, New York (1986).
- 136. W. R. SALANECK, D. T. CLARK and E. J. SAMUELSEN (ED), Science and Applications of conducting Polymers, Adam Hilger, New York (1990).
- A. G. MACDIARMID and A. J. HEEGER, NRL Memo. Rep., Proc. Mol. Electron. Devices Workshop, AD-A 05816, 208 (1981).
- 138. J.-M. Andre, J. Delhalle and J.-L. Bredas, Quantum Chemistry Aided Design of Organic Polymers, World Scientific, London (1991).
- H. KUZMANY, M. MEHRING and S. ROTH (Ed.), Electronic Properties of Conjugated Polymers, Springer, Berlin (1987).
- 140. T. A. Skothem (Ed.), Electroresponsive Molecular and Polymeric Systems, Vol. 2, Marcel Dekker, New York (1991).
- 141. T. YAMAMOTO, *Macromolecular Complexes*, (E. TSUCHIDA Ed.) p. 379, VCH Publishers, New York (1991).
- 142. S. Yoshimura, Dedensei Polymer (Conductive Polymer), Kyoritsu, Tokyo (1987).
- 143. T. YAMAMOTO and H. SAKAI, J. Mater. Sci. Lett. 4, 916 (1985).
- 144. T. YAMAMOTO, K. SANECHIKA and H. SAKAI, J. Macromolec. Sci. -Chem., A27, 1147 (1990).

- M. Kobayashi, J. Chen, T.-C. Chung, F. Moraes, A. Heeger and J. Wudl, Synth. Metals 9, 77 (1984).
- G. R. Newkome and W. W. Paudler, Contemporary Hetreocyclic Chemistry, John Wiley, New York (1982).
- 147. K. SANECHIKA, T. YAMAMOTO and A. YAMAMOTO, Polym. J. 13, 255 (1981).
- T YAMAMOTO, K. SANECHIKA, A. YAMAMOTO, M. KATADA, I. MOTOYAMA and H. SANO, Inorg. Chim. Acta 74f, 75 (1983).
- 149. T. YAMAMOTO, H. WAKAYAMA and T. KANBARA, J. Phys. Chem., in press.
- 150. G. SCHIAVON, G. ZOTTI and G. BONTEMPELI, J. Electroanal. Chem. 161, 323 (1984).
- 151. A. F. DIAZ and J. J. CASTILLO, J. Chem. Soc., Chem. Commun. 397 (1980).
- 152. A. F. Diaz, Z. I. Castillo, J. A. Logan and W.-Y. Lee, J. Electroanal. Chem. 129, 115 (1981).
- 153. E. M. GENIES and A. A. SYED, Synth. Metals 10, 21 (1984/1985).
- 154. W. WERNET and G. WEGNER, Makromolek. Chem. 188, 1465 (1987).
- 155. H. SHINOHARA, M. AIZAWA and H. SHIRAKAWA, J. Chem. Soc., Chem. Commun. 87 (1986).
- 156. J. W. ROBINSON, Handbook of Spectroscopy, Vol. 1, pp. 257-511, CRC Press (1974).
- 157. Kagaku benran, 3rd ed., p. II-578, Maruzen, Tokyo (1984).
- 158. Y. MIYAZAKI, T. KANBARA and T. YAMAMOTO, to be submitted.
- 159. K. Yoshino, S. Nakajima and R. Sugimoto, Jpn. J. Appl. Phys. 26, L2046 (1987).
- T. FUKUDA, T. KANBARA, T. YAMAMOTO, A. FUKUDA and H. TAKEZOE, 41st Annual Meet. of Polym. Soc. Jpn, May, 1992, Yokohama; Polym. Prepr. Jpn, 41 (1992).
- 161. Y. OHMORI, M. UCHIDA, K. MURO and K. YOSHINO, Jpn. J. Appl. Phys. 30, L1938 (1991).
- 162. T. KURIHARA, T. KAINO, Z.-H. ZHOU, T. KANBARA and T. YAMAMOTO, *Electron. Lett.* **28**, 681 (1992).
- 163. T. YAMAMOTO, J. Chem. Soc., Chem. Commun. 187 (1981).
- D. MacInnes Jr, M. A. Dry, P. J. Nigrey, D. P. Nairns, A. G. MacDiarmid and A. J. Heeger, J. Chem. Soc., Chem. Commun. 317 (1981).
- L. W. SHACKLETTE, R. L. ELSEMBAUMER, R. R. CHANCE, J. M. SOWA, D. M. IVORY, G. G. MILLER and R. H. BAUGHMAN, J. Chem. Soc., Chem. Commun. 361 (1982).
- 166. T. YAMAMOTO, M. ZAMA and A. YAMAMOTO, Chem. Lett. 563 (1985).
- M. HISHINUMA, M. ZAMA, K. OSAKADA and T. YAMAMOTO, *Inorg. Chim. Acta* 128, 185 (1987).
- 168. T. YAMAMOTO, M. ZAMA, M. HISHIMUMA and A. YAMAMOTO, J. Appl. Electrochem. 17, 607 (1987).
- 169. A. A. Schneider, D. E. Harney and M. J. Harney, J. Power Sources 5, 15 (1980).
- 170. T. YAMAMOTO, M. ZAMA and A. YAMAMOTO, Chem. Lett. 1577 (1984).
- 171. I. Mizoguchi and M. Ito, Kogyo Zairyo (Engineering Materials) 40, 45 (1992).
- 172. H. SHIRAISHI, Kogyo Zairyo (Engineering Materials) 40, 49 (1992).
- 173. Y. Kudo and S. Yoshimura, Kogvo Zairyo (Engineering Materials) 40, 28 (1992).
- M. KOBAYASHI, N. COLANERI, M. BOYSEI, F. WUDL and A. J. HEEGER, J. Chem. Phys. 82, 5717 (1985).
- 175. F. WUDL, M. KOBAYASHI and A. J. HEEGER, J. Org. Chem. 49, 3382 (1984).
- 176. Y. IKENOUE, Synth. Metals 35, 263 (1990).
- 177. K. KANETO, S. TAKEDA and YOSHINO, Jpn. J. Appl. Phys. 24, L553 (1985).
- 178. Y. OHMORI, M. CHIDA, K. MURO and K. YOSHINO, J. Appl. Phys. 30, L1938 (1991).
- 179. H. WAKAYAMA, T. KANBARA and T. YAMAMOTO, Polym. Prepr. Jpn 40, 3796 (1991).
- 180. S. Cao and M. S. Wrighton, J. Am. Chem. Soc. 109, 2197 (1987).
- 181. A. TSUMURA, H. KOEZUKA and T. ANDO, Appl. Phys. Lett. 49, 1210 (1986).
- 182. H. WAKAYAMA, Dissertation for M. S. Eng. to Tokyo Inst. Tech. (1992).
- 183. M. KONAGAI, K. TAKAHASHI, K. NISHIHATA, H. SHIRAKAWA and S. IKEDA, Denki Gakkai Ronbunshu 101, 103 (1981).
- 184. F. GARNIER and G. HOROWITZ, Synth. Metals 18, 693 (1987).
- T. YAMAMOTO, T. KANBARA and T. MARUYAMA, Kogyo Zairyo (Engineering Materials) 40, 23 (1992).